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# Performance and stability of femtosecond laser-irradiated $Fe_2O_3$ materials as photocatalysts for methylene blue dye discoloration

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**ABSTRACT:** The disposal of substances pollutant, such as methylene blue dye (MB), into wastewater, arouses the interest of technologies to remove these pollutants. The discoloration of MB by photocatalysis and using femtosecond laser-irradiated Fe<sub>2</sub>O<sub>3</sub> materials as photocatalysts proved to be a promising way to treat this pollutant. Here, Fe<sub>2</sub>O<sub>3</sub> obtained by the conventional hydrothermal process and heat-treatment, subsequently femtosecond laser-irradiated treatment. Materials obtained with a mixture of irradiated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub> phases revealed lower crystallinity than irradiated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> samples. The irradiation treatment and the increase in crystallinity

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were crucial to improve the performance of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> irradiated sample in the discoloration of MB, which reached 90% in 75 min of dye solution exposure under UV irradiation. The irradiation treatment proved to allow greater exposure of the particle surfaces, which coalesced and presented a greater distribution of medium size, indicating a more heterogeneous morphology in the irradiated samples. As a result, there was an increase in active sites due to the density of defects generated, which facilitated the dye degradation process.



## **1. Introduction**

Hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) and  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub> are two of the five known iron oxide polymorphs which have different crystal structures under normal conditions of temperature and pressure (CNTP) (Machala et al., 2011; Tuček et al., 2015). Under the conditions of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, the material is more thermodynamically stable and abundant within the polymorphs of  $Fe_2O_3$ (Machala et al., 2011; Sivula et al., 2011). During the period of expansion of the past Inca α-Fe<sub>2</sub>O<sub>3</sub> was found on the surface of magmatic rocks due to years of aerobic weathering processes (Gialanella et al., 2010; Sepúlveda et al., 2019), among its first uses as pigments in primitive ruptured paints in prehistoric eras (Marean et al., 2007) and more recently as dyes (Carneiro *et al.*, 2018).  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> can be obtained by thermally induced structural transformations of other oxides (Gialanella et al., 2010; Gonçalves et al., 2011) and/or other Fe<sub>2</sub>O<sub>3</sub> polymorphs (Darezereshki, 2011; Ding et al., 2007; Zboril et al., 2002), as well as from reactions between iron salts (II) and (III) (Lassoued et al., 2017; Papynov et al., 2018; Sivula et al., 2010).

ε-Fe<sub>2</sub>O<sub>3</sub> was discovered in 1934 by Forestier and Guiot-Guillain (1934), but only received the name  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> 29 years later by the researchers Schrader and Büttner (1963). Its structural characterization was performed for the first time in 1998 by Tronc et al. (1998) and in 2005 Kelm and Mader (2005), Sakurai et al. (2005) developed a model called refinement to interpret the X-ray crystal structure of  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub>, obtained in coherence with some experimental data, such as lattice parameters and electron density maps (Shabalina *et al.*, 2018). The first thin films of  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> were prepared in 2010 (Gich et al., 2010); however, their chemical stability was determined eight years later at pressures up to 27 GPa (Sans et al., 2018). E- $Fe_2O_3$  is a metastable material between the polymorphs  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and/or  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> (Tronc *et al.*, 1998). Although it is found as biogenic nanoparticles mixed with magnetite and is present in some clays, this material is not very abundant in nature. Additionally, its particles have a low surface energy, which guarantees the formation and existence of  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> (Gich *et al.*, 2007; McClean et al., 2001; Petersen et al., 1987).

Since  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is an n-type semiconductor with a gap energy between 1.9 and 2.2 eV, it is considered as a potential sunlight converter, which uses solar energy to generate clean energy, such as electricity or green fuel (e.g., H<sub>2</sub>) (Gratzel, 2001; Tamirat *et al.*, 2016). For such reason, it has been widely studied in photoelectrochemical processes (Sivula *et al.*, 2010; 2011; Trindade *et al.*, 2020). Fe<sub>2</sub>O<sub>3</sub> materials can also

be applied as carrier of drugs to treat cancer cells (Liong *et al.*, 2008; Mandriota *et al.*, 2019), lithium batteries (Gu *et al.*, 2013), gas sensors (Gou *et al.*, 2008), pigments (Carneiro *et al.*, 2018) and dyes (Ahmed *et al.*, 2013). Many studies have used Fe<sub>2</sub>O<sub>3</sub> materials as photocatalysts or adsorbents in the degradation of dyes present in industrial effluents (paper, textiles, leather) especially methylene blue (MB), which is resistant to chemical and biological treatment and can produce more toxic substances than the dye itself during its chemical decomposition (Ahmed *et al.*, 2013; Crini, 2005; Sharma *et al.*, 2011).

The versatility of Fe<sub>2</sub>O<sub>3</sub> materials is attributed to their distinct crystal structures, resulting in different physicochemical properties (Machala et al., 2011: Sakurai et al., 2009). What makes these materials versatile with interesting technological applications is the greater understanding of the association of their properties with their crystal structures, especially concerning the effects of structural order and disorder (Pottker et al., 2018). It is known that the structural modification on the surface of particles irradiated using femtosecond laser technology improves the structural and ablation properties of the material (Keller, 2003; Sugioka and Cheng, 2014). Such modification can be assigned to the creation of cations and oxygen vacancies that modulate the electronic states in the valence band of the material (Assis et al., 2020). This study evaluates the performance of thermally treated and femtosecond laser-irradiated Fe<sub>2</sub>O<sub>3</sub> samples for the discoloration of methylene blue (MB) dye under UV irradiation.

#### 2. Experimental

The Fe<sub>2</sub>O<sub>3</sub> particles were obtained by the conventional hydrothermal process, according to a procedure similar to that reported by Gou *et al.* (2008): 2 mmol of Fe(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O, 4 mmol of trisodium citrate dihydrate and 5 mmol of urea were dissolved separately in 10 mL of deionized water and subsequently dissolved and placed in a hydrothermal reactor under constant stirring. The hydrothermal process was carried out at 160 °C for 10 h. After cooling, the precipitates were washed with deionized water and alcohol, and then dried at 90 °C for 6 h. Lastly, they were heat-treated at 860 and 900 °C for 30 min.

 $Fe_2O_3$  samples were irradiated by a Ti:Sapphire laser (CPA-2001 system from Clark-MXR Inc.), at 775 nm, 150 fs (FWHM) and a repetition rate of 1 Hz. The samples were irradiated with an average power of 300 mW and positioned within the Rayleigh range. The

experiment was carried out following Assis *et al.* (2020). The nonirradiated samples were named hematite+epsilon-Fe2O3 (EPHE) and hematite-Fe2O3 (HE). After irradiation, they were denominated EPHEI and HEI, respectively.

The thermal behavior of  $Fe_2O_3$  particles obtained by the hydrothermal process were evaluated by thermogravimetry (TGA) and differential thermal analysis (DTA), and then heated up to 1300 °C in an  $O_2$  atmosphere with a flow of 50 mL min<sup>-1</sup> (NETZSCH - Cell 409).

The structural characterization of Fe<sub>2</sub>O<sub>3</sub> samples was performed by X-ray diffraction (XRD), Raman spectroscopy and scanning electron microscopy (SEM). For the XRD measurements, a diffractometer (Shimadzu) with Cu K $\alpha$  radiation ( $\lambda = 1.5406$  Å) was used, and the XRD patterns were acquired with steps of 0.02° and an angular range of 10 to 110°. The data obtained were analyzed using the Rietveld (1969) method with the aid of the general structure analysis system (GSAS) software (Von Dreele and Larson, 1994). Raman spectra were obtained in the range of 100–1000 cm<sup>-1</sup> using a Senterra spectrometer (Bruker) coupled to a 785 nm He–Ne laser excitation source with a power of 1 mW and a microscope with a 20× objective.

The morphology of the Fe<sub>2</sub>O<sub>3</sub> samples was analyzed through images obtained in a Zeiss - Supra 35 scanning electron microscope. The images were obtained by secondary electron detection (ETD, Everhart - Thornley detector) with a 10 kV incident beam. Transmission electron microscopy (TEM) images were captured in high resolution (HRTEM) using an FEI – Tecnai F20 microscope operating in the brightfield. One µm-scale scanning electron micrographs were used to construct the frequency histogram and the mean size modal distribution curve versus particle size. The count of 150 particles was inspected and the measurements were performed by the ImageJ software (Schneider et al., 2012) using the linear method.

The UV-Vis spectra were obtained over a range of 800–300 nm in diffuse reflectance mode at room temperature using a Varian Cary 5G spectrometer.

The photocatalytic activity of nonirradiated and irradiated  $Fe_2O_3$  samples was tested for the discoloration of the methylene blue dye (MB;  $[C_{16}H_{18}ClN_3S]$ ; 99.5% purity, Mallinckrodt) in relation to the exposure time under UV irradiation. The loss of MB coloration was observed by aliquots of  $Fe_2O_3$ materials dispersed in the dye solution under UV radiation at different collection times. The concentration of the aqueous solution of MB used in the photocatalysis experiments was provided by a calibration curve. The maximum absorbance ( $\lambda_{\text{maximum}}$ ) measurement of MB was performed using a UV-Vis spectrometer (V-660 Jasco). As the dye follows Beer's Law (Grasse *et al.*, 2016), the calibration curve provided the aqueous concentration of the MB index in the photocatalytic study of nonirradiated and irradiated Fe<sub>2</sub>O<sub>3</sub> materials.

The experiments were carried out by dispersing 50 mg of the sample in 50 mL of MB solution in an ultrasound bath inside an open reactor with a controlled temperature of 20 °C and water circulation. The sample-dye adsorption process was performed by stirring this dispersion in the dark for 30 min. The photocatalytic system was obtained by illuminating this dispersion with six UV lamps (TUV Philips, 15 W with maximum intensity of 254 nm). Aliquots were removed at time zero and other predetermined times (5, 10, 20, 30, 45, 60 and 75 min), the absorbance changes were measured by a spectrophotometer (V-660 Jasco), while the photocatalytic stability of the sample with the best performance was obtained from the recycling test results.

# 3. Results and discussion

There are four thermal events according to the TGA and differential thermal DTA curves of the Fe<sub>2</sub>O<sub>3</sub> sample obtained by the conventional hydrothermal process, as showed in Fig. 1. The TGA curve demonstrates that the thermal stability of the sample occurred at 500 °C. The processes with energy absorption were identified at 110 °C, which corresponds to the sample dehvdration, and at 650 °C. indicating that above this temperature hematite presents paramagnetic behavior (Liu et al., 1997). At 243 and 860 °C, two exothermic events were observed, the first due to the release of gases and organic impurities from the process of particle obtention, and the second attributed to the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub>  $\rightarrow \alpha$ -Fe<sub>2</sub>O<sub>3</sub> phase transition (Dézsi and Coey, 1973). From this result, it was possible to perform the heat-treatment of the  $Fe_2O_3$ samples.

The vibrational modes of Raman showing EPHE and EPHEI were identified as shown in Fig. 2. The vibrational modes of single-phase  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> samples phase are not identified in the literature yet. The determination of the wavenumbers referring to the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> phase was performed from a Lorentzian fit of the Raman scattering spectrum bands. Thus, it was possible to distinguish the vibrational modes of the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> phases (López-Sánchez *et al.*, 2016). The vibrational modes agree with the vibrational modes reported by López-Sánchez *et al.* (2016), the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> phase has 117 active Raman vibrational modes: 29A<sub>1</sub> + 30A<sub>2</sub> + 29B<sub>1</sub> + 29B<sub>2</sub>, considering that the same has space group Pna21. Sample EPHE (Fig. 2a) showed nine active modes of the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> phase in wavenumbers in the range 120 to 600 cm<sup>-1</sup>.

According to the literature (Faria *et al.*, 1997),  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> exhibits seven modes of active phonons (2A<sub>1g</sub> + 5E<sub>g</sub>) allowed in Raman. Sample EPHE shows two modes in 226 and 496 cm<sup>-1</sup>, attributed to A<sub>1g</sub>, and others in 245, 293, 410 and 610 cm<sup>-1</sup>, assigned to E<sub>g</sub>.

The changes observed in the Raman vibrational modes of the irradiated samples (Fig. 2b) are a consequence of the irradiation treatment, which provided an increase in the structural disorder within the crystal lattice of samples EPHEI and HEI. Compared to the nonirradiated samples, the high full-width at half-maximum (FWHM) values of the Raman bands, and the absence of active vibrational modes of the irradiated samples (Tab. 1) indicated that the irradiation treatment caused a distortion in the crystal lattice of the samples and an increase in the defect

density, leading to the breaking of Fe–O bonds. These results contribute to changes in the physicochemical properties of the irradiated samples.



**Figure 1.** Differential thermal curve (olive line) and TGA curve (magenta line) of  $Fe_2O_3$  sample obtained by the conventional hydrothermal process.



Figure 2. Raman spectra for (a) nonitradiated samples and (b) madiated samp

108

X-ray diffraction patterns of samples EPHE and EPHEI (Fig. 3) presented peaks indexed to the inorganic crystal structure database (ICSD) number 415250 (Kelm and Mader, 2005; Sakurai *et al.*, 2005) and 15840 (Blake *et al.*, 1966), and corresponding to  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> phases, respectively, characterizing these materials as multiphase. These phases have an orthorhombic (Pna 21) and rhombohedral (R-3cH)

structure with eight and six molecular formulas per unit cell (Z = 8 and 6), respectively. On the other hand, samples HE and HEI are formed by  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and are considered monophasic. This preferential formation is due to the increase in temperature to 900 °C during the heat-treatment when compared to samples EPHE and EPHEI.

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Wavenumber	FWHM (cm <sup>-1</sup> ) FPHF	Wavenumber	FWHM	Wavenumber	FWHM	Wavenumber	FWHM
				(CIII ) IIIE	(cm) me		(cm) mei
$123.4 \pm 0.1$	11.3	$122.7 \pm 0.3$	13.8	-	-	-	-
$147.9\pm0.2$	9.6	$148.2\pm0.2$	13.3	-	-	-	-
$172.1 \pm 0.3$	10.5	$172.0\pm0.3$	11.1	-	-	-	-
$226.1 \pm 0.3$	6.8	$225.7\pm0.3$	2.8	$225.8\pm0.1$	8.9	$224.9\pm0.1$	8.7
$244.9\pm0.2$	6.2	-	-	$244.1\pm0.2$	11.5	$245 \pm 2$	45.0
$293.8\pm0.2$	13.8	$293.9\pm0.3$	4.9	$292.8\pm0.1$	14.7	$292.1 \pm 0.1$	15.7
$326.7 \pm 0.3$	7.1	$325.4 \pm 0.2$	6.0	-	-	-	-
$349.6 \pm 0.2$	6.2	$350.6 \pm 0.1$	4.7	-	-	-	-
$375.5 \pm 0.2$	12.2	$373.9 \pm 0.1$	5.1	-	-	-	-
$409.1 \pm 0.3$	16.6	$409.3 \pm 0.3$	6.9	$409.3 \pm 0.1$	15.6	$408.9\pm0.1$	19.0
$443.0\pm0.3$	8.4	-	-	-	-	-	-
$560.9\pm0.2$	7.8	-	-	-	-	-	-
$590.0\pm0.2$	6.7	-	-	-	-	-	-
$495.3 \pm 0.3$	32.5	$494.3 \pm 0.2$	10.4	$494.0 \pm 2.1$	25.2	$493.0\pm0.3$	22.3
$610.1 \pm 0.2$	9.9	-	-	$610.5 \pm 0.2$	12.2	$608.7 \pm 0.1$	19.1

Table 1. Wavenumbers (cm<sup>-1</sup>) and full widths at half maximum (FWHM) for nonirradiated and irradiated samples



**Figure 3.** X-ray diffraction patterns for the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> of the samples heat-treated at 860 and 900 °C for 30 min. In (a) nonirradiated samples and (b) irradiated samples.

The results of structural refinement by the Rietveld (1969) method are shown in Tab. 2 and Fig. 4. Table 2 presents the lattice parameters and the percentage of phases of the samples before and after the irradiation treatment, whereas Fig. 4 shows typical Rietveld refinement graphs. It is possible to observe that the diffraction patterns calculated and obtained in Fig. 3 are related to high structural order or long-range and well-defined peaks.

The results of two lattice parameters presented in Tab. 2 are consistent with the previous studies of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> and  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub> phases (Blake *et al.*, 1966; Kelm and Mader, 2005; Sakurai *et al.*, 2005). In this study, EPHE and EPHEI showed 62% of the  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub> phase,

while HE exhibited crystallized  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> as the preferential phase, which is indicative of the increase in temperature without heat-treatment. These results refer to the nonlinearity of the adjustment parameter (Rw %) for different processes of crystallization, solubilization and recrystallization. Microstrain results from irradiated samples (EPHEI and HEI) revealed higher values than those shown by nonirradiated samples (EPHE and HE), demonstrating that the irradiation treatment favored or increased the density of defects and stress in the crystalline lattice of the materials.

Sample			EPHE	EPHEI	HE	HEI	<b>ε-Fe</b> 2 <b>O</b> 3 <sup>*</sup>	<b>α-Fe<sub>2</sub>O<sub>3</sub>**</b>
a-Fe <sub>2</sub> O <sub>3</sub>	Lattice parameters (Å)	a = b	5.035	5.037	5.040	5.035	-	5.038
		с	13.733	13.738	13.758	13.748	-	13.772
	V (Å <sup>3</sup> )		301.50	301.93	302.70	301.89	-	302.72
	Crystallite size (Å)		0.45	0.40	0.84	0.73	-	-
	Microstrain		1282.7	2215.5	921.9	1390.9	-	-
	% Wt fraction		0.376	0.379	1.000	1.000	-	
8-Fe <sub>2</sub> O <sub>3</sub>	Lattice parameters (Å)	а	5.091	5.091	-	-	5.071	-
		b	8.781	8.789	-	-	8.736	-
		с	9.465	9.470	-	-	9.418	-
	V (Å <sup>3</sup> )		423.09	423.74	-	-	417.25	-
	Crystallite size (Å)		0.25	0.25	-	-	-	-
	Microstrain		-599.4	-1741.8	-	-	-	
	% Wt fraction		0.624	0.621	-	-	-	-
Rw%			9.414	8.943	-	-	-	-

**Table 2.** Parameters obtained from Rietveld refinements of Fe<sub>2</sub>O<sub>3</sub> samples.



Figure 4. Rietveld refinement plot of nonirradiated in (a, c) and (b, d) irradiated samples.

Figure 5 illustrates the UV-Vis spectra in diffuse reflectance mode in the 250–800 nm range for the nonirradiated and irradiated samples. The absorbance close to the 570 nm region shows that these samples have application as absorbers (Sarma *et al.*, 2020).

The calculations performed to determine the energy of the samples in the forbidden band ( $E_{gap}$ ) were obtained according to Wood and Tauc (1972) and Chen *et al.* (2013). The indirect transitions of the samples resulted in  $E_{gap}$  values consistent with the literature (Pandey *et al.*, 2014). Compared to the nonirradiated samples, EPHEI and HEI showed a reduction in their  $E_{gap}$  value, indicating that the irradiation treatment caused the samples to present a more conductive behavior with greater structural disorder at medium-range (Pinatti *et al.*, 2020). It can be attributed to the presence of electronic levels within the forbidden band, which are associated with crystalline lattice disturbances. These results corroborate those obtained in the Raman and XRD analyses.



Figure 5. UV-Vis spectra in (a, b) and energy gap. In (c, e) nonirradiated samples and (d, f) irradiated samples.

The electron micrographs in Fig. 6 reveal that sample EPHE has a heterogeneous morphology with clusters in the shape of irregular plates and equiaxed structures on the plates. In turn, HE showed that the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> particles have a homogeneous morphology with clusters of irregular plaques. The same change was

observed in EPHEI and HEI, even though the irradiation treatment allowed the formation of equiaxed particles on the agglomerated structures, including, in some regions of the samples, the phenomenon of coalescence resulting from diffusion processes of matter during irradiation (Anastasiou *et al.*, 2016).



Figure 6. Micrographs of the nonirradiated in (a, c) and (b, d) irradiated samples.

The frequency histograms in Fig. 7 reveal an average size of 41 and 40 nm for the particles of samples EPHE and HE, respectively, and 49 and 54 nm for the irradiated samples EPHEI and HEI, respectively. It can be noted that the mean size distribution curve of the EPHE and EPHEI particles was very similar to each other. On the other hand, the mean size distribution curve of HEI became much broader than that of HE, indicating that there was an increase in the morphological heterogeneity.

Images obtained from high-resolution micrographs (HRTEM) of the nonirradiated and irradiated samples are displayed in Fig. 8. The crystallinity of the samples made it possible to obtain the indexation of the Fe<sub>2</sub>O<sub>3</sub> phases. The insets show that the nonirradiated and irradiated samples crystallized in the  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> phases, corroborating the results of the Raman spectra and XDR patterns. The irradiation treatment increased the defect density and avoided the  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub> to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> phase transition in sample EPHEI.



112

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Figure 7. Frequency histograms of the nonirradiated in (a, c) and (b, d) irradiated samples.



Figure 8. Transmission electron microscopy imagens of the nonirradiated in (a, c) and (b, d) irradiated samples.

113

The MB calibration curve (Fig. 9) was constructed from the solutions of the standard MB at pH = 7. The concentration of aqueous solutions of MB was estimated by measuring the absorbance at maximum wavelengths of MB ( $\lambda_{maximum}$ = 668 nm) (Sheng and Mat Yunus, 2005). The adjustment of the experimental data was given by a straight line with a regression coefficient of R<sup>2</sup> = 0.9990. The calibration curve allowed us to determine that the concentration of 52.33 mg L<sup>-1</sup> of MB had a maximum absorbance value of 1.0067, which is in agreement with the value found by Al-Rubayee *et al.* (2016). This was the concentration used in the photocatalysis test.



**Figure 9.** Calibration curve methylene blue at different concentrations 5–80 mg.L<sup>-1</sup>.

The photocatalytic activity of nonirradiated and irradiated  $Fe_2O_3$  samples was compared when MB was subjected to UV irradiation, in the absence of

photocatalysts and under the same experimental conditions. Figure 10a shows the relationship among the equilibrium adsorption concentration. the concentration after irradiation  $(C_N/C_0)$  and the irradiation time. The MB sample without catalyst showed much lower discoloration than catalyzed samples. The discoloration of MB in 75 min photocatalyzed by HEI was 90%. However, HE showed a similar discoloration profile in the absence of a catalyst. In relation to the dye adsorption, samples EPHE and EPHEI reached values of 60 and 35% in the dark, respectively. In contrast, HE ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>) adsorbed 20% of the dye within 30 min of exposure in the dark. Allawi et al. (2020) studied the adsorption process of 100 mg of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> catalyst in a 20 mg L<sup>-1</sup> MB solution at pH 7.6. They observed that in 6 min of exposure in the dark, the dye adsorption was  $\sim 10$  %. Considering the results achieved by samples EPHE and EPHEI, it is possible to then infer that the presence of the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> phase significantly influenced the MB adsorption process. During the photocatalysis of EPHE and EPHEI, the discoloration was 8 and 15%, respectively, which is in accordance with the results obtained by Ahmed et al. (2013). These findings show that samples EPHE and EPHEI behave as MB adsorbent materials, being promising for MB photocatalysis. Regarding the photocatalytic activity of HEI, the recycling test was performed following the procedure described Trench et al. (2018). The effect of photocatalytic activity of each cycle is shown in Fig. 10c. It is possible to observe that the photocatalytic activity of this sample decreased over the cycles, which can be attributed to the loss of material due to washing processes for dye removal. On the other hand, such sample (HEI) showed good stability as a photocatalyst.



**Figure 10.** Photocatalytic discoloration of MB (52.33 mg  $L^{-1}$ ). In (a) without catalyst and Fe<sub>2</sub>O<sub>3</sub> nonirradiated and irradiated materials and (b) Determination of the rate constant by log plot and (c) Recycle test of HEI sample.

For the MB dye, the solution concentration was considered very low. In order to quantitatively compare the obtained results, a kinetic study was carried out using a pseudo first-order reaction from the Langmuir-Hinshelwood model, as showed Eq. 1 (Guettaï and Amar, 2005):

$$\mathbf{k}'\mathbf{t} = -\ln\left(\frac{c_N}{c_0}\right) \tag{1}$$

where k' and t represent the constant reaction rate and the irradiation time, respectively. Figure 10b shows the kinetic behavior of MB discoloration. The reaction constants found were 2.67×10<sup>-2</sup>, 1.39×10<sup>-3</sup>, 1.80×10<sup>-3</sup>, 1.87×10<sup>-3</sup>, 3.52×10<sup>-4</sup> min<sup>-1</sup> for HEI, EPHEI, HE, EPHE and the sample without catalyst, respectively. In comparison with the sample without catalyst. EPHE, EPHEI and HE showed an increase in the reaction speed in one order of magnitude, while HEI exhibited a constant speed with two orders of magnitude.

Figures 6 and 8 illustrate the morphology of the nonirradiated samples (EPHE and HE) under thermal treatment. It could be observed that the femtosecond laser-irradiation treatment led to the formation of different morphologies in EPHEI and HEI that interfere with the absorption and electron transfer processes, presenting different photocatalytic activities (Vu et al., 2019). The lower photocatalytic activity was due to the reduction of active reaction sites in the studied samples (Huang et al., 2015). The irradiation treatment directly affected the surface of the particles, causing an increase in the density of surface defects, thus improving the photocatalytic properties of these samples.

This result is very significant compared to the development of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> materials as a photocatalyst for MB discoloration at room temperature, which reached a value of 78% in 6 h of UV irradiation (Vu et al., 2019). In fact, the irradiation treatment proved to be very efficient to improve the photocatalytic activity of sample HEI, composed solely of α-Fe<sub>2</sub>O<sub>3</sub>. This improvement was also attributed to the defects generated on the surface of the particles (Vu et al., 2019).

The  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> present in the studied samples has short diffusion length of the minority carriers (holes) (Wheeler et al., 2012) and low conductivity of the majority carriers (electrons) (Kennedy and Frese Junior, 1978), resulting in difficulty in separating the photogenerated electron-hole pairs, and consequently influencing the photocatalytic activity results of these materials. On the other hand,  $\epsilon$ -Fe<sub>2</sub>O<sub>3</sub> is a material difficult to be obtained in isolation (Danno et al., 2013; Dézsi and Coey, 1973; Shanenkov et al., 2019; Wang et al., 2019). For such reason, its attention is more focused on its interesting magnetic properties, as the anisotropic particles in this material can lead to the formation of a single magnetic domain, generating a large coercive field compared to other Fe<sub>2</sub>O<sub>3</sub> polymorphs, and thus having advanced magnetic applications (Jin et al., 2004; Machala et al., 2011). However, in this study it was very interesting to investigate the photocatalytic properties of this material against α-Fe<sub>2</sub>O<sub>3</sub>.

Since the photocatalytic process of HEI occurred by the transfer of surface charge, there was a consequent increase in defect density (Huang et al., 2015). Such sample absorbed energy of the photon generated by UV irradiation to form charge carriers (electron-hole pairs) that participate in the photooxidation on the sample surface. Minority carriers form very reactive hydroxy radicals (·OH), which are the main active species used to decolor MB (Zhao et al., 2004). In contrast, majority carriers captured by dissolved oxygen in the MB solution form superoxide radicals anions  $(\cdot O_2^{-})$  and/or peroxide radicals, which have the ability to break down MB molecules until mineralizing them into CO<sub>2</sub> and H<sub>2</sub>O (Vadivel *et al.*, 2014). In turn,  $\cdot O_2^-$  reacts with minority carriers of the sample, forming peroxides and releasing oxygen and/or active OH radicals and peroxides, partially contributing to the degradation of MB (Huang et al., 2015).

#### 4. Conclusions

The optical properties of femtosecond laserirradiated  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> materials showed a reduction in their band gap energy values and exhibited a more conductive behavior than nonirradiated samples due to the presence of electronic levels within the band gap range generated by the resulting defect density caused by irradiation.

According to the XRD patterns, the structural characterization showed that the high crystallinity of samples HE compared to EPHE influenced the increase in their photocatalytic activity due to the presence of the  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> phase. On the other hand, the femtosecond laser irradiation treatment improved the photocatalytic properties of EPHEI compared to the nonirradiated sample (EPHE). A significant reduction in the Raman modes was observed for irradiated samples, indicating that their structural disorder at medium range was a result from the increase in the density of defects generated on the surface of their particles. Therefore, it was proven that sample HEI presented the best photocatalytic property, which was attributed to the

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115

synergistic effect of the high crystallinity of the  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> phase and the generation of defects caused by the irradiation treatment.

The morphological aspect of the samples revealed agglomerated particles that impaired the improvement of the photocatalytic properties of the samples. Therefore, samples EPHE, HE and EPHEI behaved as adsorbent materials for MB discoloration. These materials have the potential to be used together with photocatalysts to enhance MB degradation in a short period of dye exposure. The mean particle size distribution curve of HEI was wider than that of HE, indicating increase in the morphological an heterogeneity of the irradiated sample, which consequently favored the increase in its catalytic activity. The phase indexing in the HRTEM images showed that the irradiation treatment was not energetically favorable to the transformation of  $\varepsilon$ -Fe<sub>2</sub>O<sub>3</sub> to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, but sufficiently effective for the generation of surface defects in the particles of the irradiated samples (HEI and EPHEI).

# Authors' contribution

**Conceptualization:** Souza, J. C.; Martins, T. A.; Oliveira, R. C.; Sambrano, J. R.; Boni, L.; Mendonça, C. R.; Leite, E. R.; Longo, E.

Data curation: Not applicable.

**Formal Analysis:** Souza, J. C.; Martins, T. A.; Oliveira, R. C.; Boni, L.

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**Resources:** Souza, J. C.; Martins, T. A.; Oliveira, R. C.; Boni, L.; Mendonça, C. R.; Leite, E. R.; Longo, E. **Software:** Not applicable

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**Visualization:** Souza, J. C.; Martins, T. A.; Oliveira, R. C.; Sambrano, J. R Mendonça, C. R.; Boni, L.

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# Data availability statement

All data sets were generated or analyzed in the current study

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116

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117

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