

## **INVESTIGATION OF DYE-SENSITIZED SOLAR CELL PERFORMANCE BASED ON VERTICALLY ALIGNED TiO<sub>2</sub> NANOWIRE PHOTOANODE \***

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**Abstract.** *In this work, we present our results related to the development of Dye-Sensitized Solar Cells (DSSCs) based on vertically aligned TiO<sub>2</sub>-nanowire (NW) and Ag nanoparticle (NP) assisted vertically aligned TiO<sub>2</sub>-NW (TAT) photoanode fabricated by the glancing angle deposition (GLAD) technique on fluorine doped thin oxide (FTO) substrates. The scanning electron microscopy (SEM) analysis reveals that the Ag-NP assisted vertically aligned TiO<sub>2</sub>-NW photoanode was successfully deposited on FTO substrates. The average length and diameter of the NW have been measured to be ~ 350 nm and ~ 90 - 100 nm, respectively. Moreover, transmission electron microscopy (TEM) and X-ray diffraction (XRD) manifest the presence of small crystals of TiO<sub>2</sub> and Ag. Further, the absorption spectrum analysis reveals that the incorporation of Ag-NP in TiO<sub>2</sub>-NW increases absorption in the visible region, but decreases the efficiency of the cell after the incorporation of the nanoparticle. The calculated bandgap of the annealed Ag-NP (30 nm) assisted TiO<sub>2</sub>-NW (TAT@30nm) sample from the photoluminescence (PL) graph is ~ 3.12 eV. Finally, it is observed that the TiO<sub>2</sub>-NW based DSSC device shows better performance in terms of photo conversion efficiency (PCE) compared to the TAT@30nm photoanode based device, with an efficiency of ~0.61 % from the former and ~ 0.24 % from the latter. This reduction in the efficiency of TAT@30nm based devices is due to the larger size of Ag-NP, in which the nanoparticle acts as an electron sink and acts as a blocking layer.*

**Key words:** DSSCs, e-beam, nanowire, nanoparticle, TiO<sub>2</sub>

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## 1. INTRODUCTION

The number of natural disasters is rising daily due to the rapid climate change in the last few decades, directly or indirectly caused by carbon emissions from fossil fuels and the destruction of forest areas. Therefore, it is important to develop a sustainable energy conversion device to overcome the issue of increasing energy demand. Fossil fuels are one of the main sources of energy that will eventually run out, which will have an effect on the environment and the ecosystem in the area. As a result, scientists are always working to improve renewable energy sources like hydroelectricity, solar energy and wind energy. Since a significant amount of sunlight penetrates the earth's surface, solar cells, among other energy sources, play a significant part in the generation of electrical energy. Lots of design and development have been done on photovoltaic technology to maximize the conversion efficiency of sunlight. The most commonly used solar cell materials are silicon, perovskites, graphene, III-V nitrides and organic dyes [1-6]. Among these, Dye-Sensitized Solar Cells (DSSCs) became attractive after O'Regan and Gratzel's reported the outstanding properties of DSSCs like the multicolor option, easy integration into building architecture, ease of fabrication, low cost and affordability [7]. Similar to how plant chlorophyll performs photosynthesis, this solar cell's operation relies on the photo-electrochemical reaction, in which the dye molecule acts as a molecular electron pump by trapping the light. When light falls on the surface of the cell, the excited dye molecule is oxidized and transferring those excited electrons into the conduction band of a wide bandgap semiconductor, such as nanostructured  $\text{TiO}_2$ . The excited electron in the  $\text{TiO}_2$  nanostructure is then transported to the counter electrode by the process of diffusion through the external circuit. Again, the oxidized dye molecule present inside the cell is regenerated from iodine present in the redox electrolyte medium and further regenerated from iodine by reduction of triiodide on the counter electrode [8]. The four different modules that make up DSSCs are photoanode, dye, electrolyte, and counter electrode. Among these, the photoanode is crucial to the process of photon conversion and the dye sensitizer influences how well the DSSCs work. The sensitizer/dye should possess broad and strong absorption from the visible to the near region. Ruthenium compound is the most commonly used efficient and stable dye. Even though these dyes have some disadvantage compared to Eosin-Y and porphyrin, they have excellent electron injection, higher absorption in the visible range, good stability and efficient charge transfer, thereby giving the highest efficiency [9, 10].

Moreover,  $\text{TiO}_2$  has a high bandgap, resistance to photo corrosion, and nontoxicity compared to other metal oxides like  $\text{ZnO}_2$ ,  $\text{SnO}_2$ ,  $\text{Cu}_2\text{O}$ ,  $\text{WO}_3$  and  $\text{In}_2\text{O}_3$  [11-12]. Naturally,  $\text{TiO}_2$  crystal, which belongs to the transition metal oxides, can assume any of the three forms, i.e., anatase, rutile or brookite [13-14]. Anatase  $\text{TiO}_2$  is mainly employed to create photoanodes for DSSCs because of its higher charge transport and stability. The power conversion efficiency (PCE) of the DSSCs is significantly impacted by the form and size of the  $\text{TiO}_2$  nanoparticle.  $\text{TiO}_2$  is a chemically inert substance because it has a bandgap of  $\sim 3.2$  eV or less and does not induce chemical reactions in the absence of light. Due to scaling laws, the chemical and physical properties of nanomaterials change as their geometrical dimension decreases [15]. So, there has been recent progress in the synthesis of  $\text{TiO}_2$  nanomaterial like nanorods (NRs), nanowires (NWs) and nanotubes (NTs), which possess different properties because of the different synthesis techniques, unique nanostructure, and high surface-to-volume ratio that enhance the delocalized carrier charge particle, thereby

increasing the charge transportation. This 1D nanostructure can be used in designing photoanodes for the DSSCs application, which can enhance the efficiency through rapid electron transport.

Nanomaterial synthesis can be done in different ways, like the sol-gel method, the hydrothermal method, chemical vapour deposition (CVD) and physical vapour deposition (PVD) techniques, etc. [16-17]. Furthermore, TiO<sub>2</sub>-NW enhanced the performance of the energy-sensing because of more reaction sites due to the high surface area and larger extension of the depletion region. In addition to that, TiO<sub>2</sub>-NW has confined conductive channels which can reduce charge recombination, hence enhancing charge transportation as compared to other bulk structures. Yang *et al.* reported that porous TiO<sub>2</sub> TF was deposited by glancing angle deposition (GLAD) using e-beam deposition. Also, it was reported that TiO<sub>2</sub> film has the largest internal surface area, which enhances the dye absorption of the DSSCs [18]. Wong *et al.* reported that TiO<sub>2</sub> photoanodes were prepared using the e-beam technology. It is also observed that the efficiency of 6.1% was achieved at an inclined GLAD angle of 73° which improves the light trapping nature of the TiO<sub>2</sub> photoanode as it is a columnar structure [19]. The highest reported PCE of DSSCs is ~ 14.2 %, which is fabricated using the chemical process on screen-printed TiO<sub>2</sub> film [20]. Even though the efficiency of the DSSCs is much lower than that of Si solar cells, they have remarkable performance under low light intensity, which can be used in indoor lighting. So, in this work, the GLAD method was used to grow vertically aligned TiO<sub>2</sub>-NW as photoanodes on fluorine-doped thin oxide (FTO) for DSSC application without using any catalyst. Furthermore, it should be noted that the GLAD can be used to precisely control the shape, size, and thickness of the nanostructure [21]. The vertical TiO<sub>2</sub> nanostructure achieved from GALD deposition enhances the efficiency by shortening the electron pathway through the vertical TiO<sub>2</sub>-NW. Further, an attempt has been made to put Ag metal nanoparticles in the middle of the TiO<sub>2</sub>-NW to enhance the photon absorption through surface plasmon resonance (SPR). The SPR effect mainly depends on the type of metal used, its shape and the size of the metal nanoparticle. Both Ag and Au exhibit a strong SPR effect in the visible region. However, the cost of the Ag metal is comparatively lower as compared to Au. Again, Ag-NP is highly stable and can withstand corrosion and less oxidized [22]. It is noteworthy to mention that the Ag nanoparticle can be used in various applications like supercapacitors, biosensors and other optoelectronic applications, etc [23-25].

Therefore, an effort is made to develop DSSC based on TiO<sub>2</sub>-NW and Ag-NP embedded vertical TiO<sub>2</sub>-NW photoanodes deposited by the GLAD method on FTO substrates for the DSSC application. The samples were analyzed using scanning electron microscopy (SEM), Transmission Electron Microscopy (TEM) and X-ray diffraction (XRD) (RigaKu Ultima IV, CuK $\alpha$  radiation,  $k = 0.1540$ ) analysis for morphology and structural analysis, respectively. Finally, the performance of two types of DSSCs, i.e., TiO<sub>2</sub>-NW and Ag-NP assisted TiO<sub>2</sub>-NW photoanode based devices, is analyzed.

## 2. EXPERIMENTAL DETAILS

### 2.1. Materials

Both the TiO<sub>2</sub> and Ag (both 99.999% pure) were procured from Tecnisco Advanced Materials Pte Ltd, Singapore. N719 Ruthenium dye sensitizer (95% pure) was purchased from SRL Pvt. Ltd, FTO/glass (12-14  $\Omega/\text{cm}^2$ ) from MTI Corporation, USA and Iodolyte

HI-30 electrolyte were purchased from Solaronix, Switzerland. For the deposition of the TiO<sub>2</sub> and Ag, the material is loaded into the crucible and put into the evaporation chamber. Before creating the vacuum, the chamber was cleaned properly by applying acetone. Further, the vacuum is created inside the chamber and the samples are inclined at 81° during the NW and NP deposition.

## 2.2. Photoanode preparation

FTO glass substrates having a resistivity of ~ 12-14 ohm/cm<sup>2</sup> were properly cleaned sequentially by rinsing them in deionized water (DI) (Oxford Lab Fine Chem LLP (CAS No. 7732-18-5)) for 1 minute each and drying them in the open air for 5 minutes before putting them inside the chamber. The vertically oriented TiO<sub>2</sub>-NW and Ag-NP assisted vertically aligned TiO<sub>2</sub>-NW (TAT) are deposited on FTO coated glass substrates by the GLAD using an e-beam evaporator (Model No. Smart Coat 3.0, HHV India). This GLAD mechanism, which is installed inside the chamber, allows the change of the angle by moving the axis of it. In our previous work, the details of the fabrication process of TiO<sub>2</sub>-NW and TAT samples were discussed [21]. Here, the process is explained in brief. The samples are kept at an inclined angle of 81° and rotated at 30 rpm to form a vertically aligned TiO<sub>2</sub> nanostructure. In the first round of deposition, TiO<sub>2</sub>-NW (350 nm) samples were deposited on an FTO-coated glass (1 cm x 1 cm) substrate by employing the GLAD method. For another group of samples, TiO<sub>2</sub>-NW (175 nm) was initially deposited on glass (1 cm x 1 cm). Further, Ag-NP (30 nm) was deposited above the TiO<sub>2</sub>-NW (175 nm). Again, TiO<sub>2</sub>-NW (175 nm) was deposited above the Ag-NP (30 nm)/TiO<sub>2</sub>-NW (175 nm). Finally, we achieved the staking of TiO<sub>2</sub>-NW (175 nm)/Ag-NP (30 nm)/TiO<sub>2</sub>-NW (175 nm) (TAT@30nm) samples by employing the GLAD method. For every TiO<sub>2</sub> (30 nm) deposition, deposition was done for 12 minutes at a rate of ~ 0.6 Å/sec and the Ag (30 nm) deposition rate was kept constant at 0.8 Å/sec for 7 minutes. The deposition rate and thickness of the deposited film were monitored through a digital thickness monitoring system in all the deposition process to control the film thickness precisely. Similarly, vertically aligned TiO<sub>2</sub>-NW with Ag 60 nm (TAT@60nm) and Ag 90 nm (TAT@90nm) samples are prepared using the same process and parameters. All these processes are performed under high vacuum conditions of ~ 2 x 10<sup>-5</sup> mbar. The pressure of the chamber was maintained at ~ 6 x 10<sup>-6</sup> mbar before the start of deposition. However, during the deposition, the pressure drops to ~ 2 x 10<sup>-5</sup> mbar. Further, the photoanode samples for DSSCs fabrication are annealed at 500 °C for 3 hours. The samples were cooled down slowly and processed for dye loading.

## 2.3. Dye Preparation and Counter Electrode Preparation

6mg of N719 (95%, SRL Pvt. Ltd.) dye salt powder is mixed with a 0.5 mM concentration of ethanol using a vortex (Ependorf MixMate) at 200 rpm for 20 minutes to make 10 ml of N719 dye solution. The resulting dye solution is kept for 1 day for stabilization, as shown in Fig. 1 (inset). Fig. 1 shows the absorption peak of the N719 sample being measured using a UV-Vis spectrophotometer (AN-UV-6500N ANTech), which reveals four bands at ~ 504 nm, ~ 376 nm and ~ 308 nm, with a shoulder peak at ~ 252 nm. The two peaks in the visible band are attributed to metal-to-ligand charge transfer (MLCT).

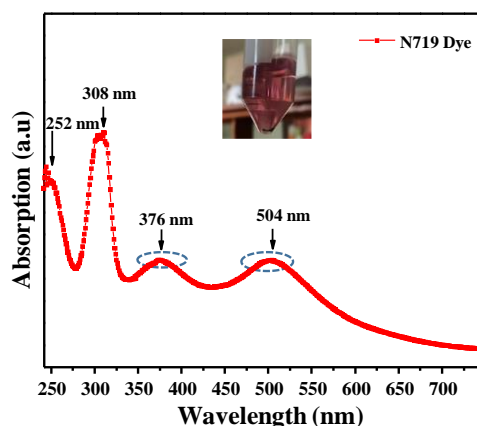


Fig. 1 Shows optical absorption spectrum of the N719 dye sample

The TiO<sub>2</sub>-NW and TAT coated FTO glass samples are immersed in the N719 dye for 24 hours, which is kept at room temperature in a dark room. To remove the excess dye, the TiO<sub>2</sub> photoanode sample is washed gently with ethanol after taking it out of the dye solution and dried for 3 minutes in the open air. Again, Plastisol T/SP paste from the solaronix was coated on the FTO substrate using the doctor-blade technique for making the counter electrode (CE). Here, the 3M scotch tape covers all four edges of the sample by keeping a 2 x 2 cm<sup>2</sup> space at the centre of the FTO glass. This sample is placed in the furnace for annealing at 450°C for 1 hour, which will activate the Pt particles.

#### 2.4. Fabrication of DSSCs

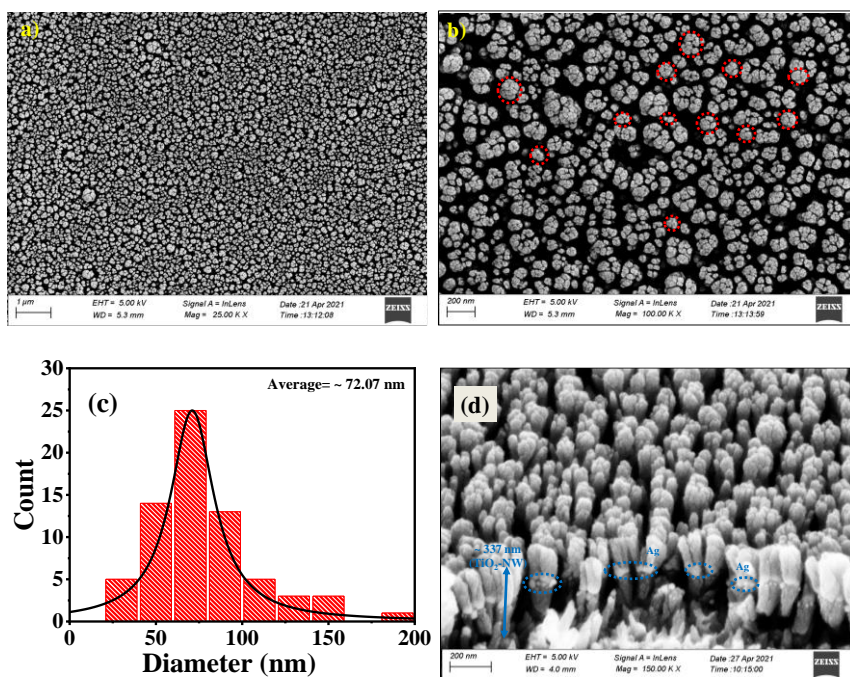
The dye-sensitized TiO<sub>2</sub> photoanode and Pt-coated counter electrode were preheated at 100 °C before being sandwiched together. This pretreated process will remove the moisture present on the surface of the photoanode and counter electrode. Further, the Pt activated counter electrode is sandwiched and sealed with the TiO<sub>2</sub> photoanode by using a paper clip to complete the DSSCs module. Lastly, the electrolyte solution was introduced in between the electrodes by capillary action.

### 3. RESULTS AND DISCUSSION

#### 3.1. SEM Analysis

The morphology of the as-deposited TAT@30nm sample was analyzed using a SEM instrument, as shown in Fig. 2 which shows the successful deposition of TAT@30nm nanowires. The magnified SEM image of TAT@30nm sample is shown in Fig. 2(b). The larger diameter nanowires indicated by dotted circle, shown in Fig. 2(b), are built by cluster formation through shadowing effects during the deposition [26]. The average top diameter of the TAT@30nm was measured and calculated from the magnified SEM image and found to be ~ 72 nm, as shown in Fig. 2(c). Fig. 2(d) shows the cross-sectional image of TAT@30nm. This image proves that vertical TAT@30nm is successfully grown onto the FTO substrate by employing the GLAD technique. It also reveals the

presence of Ag-NPs, which are indicated by blue dotted circles in the middle of the NWs. The height of the TAT nanowire is  $\sim 337$  nm.



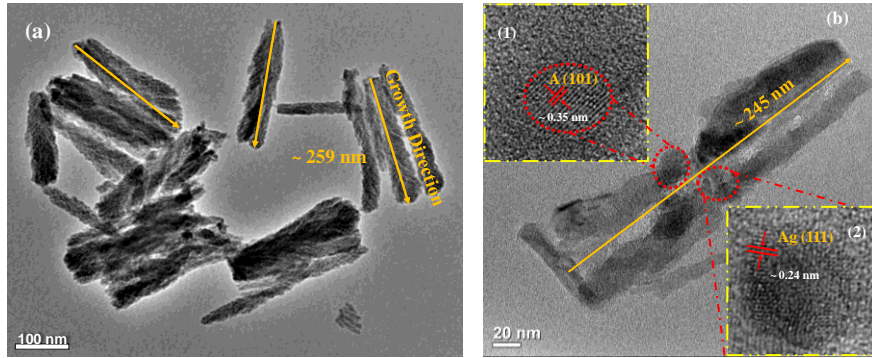
**Fig. 2** (a) The SEM image of the as-deposited TAT@30nm, (b) Represent the magnified image showing the porous nature of the sample and (c) Showing the calculated average diameter, (d) a cross-sectional image of the TAT@30nm sample

These vertical nanostructures enhanced the efficiency of the DSSC solar cell by enhancing the surface area of the active layer as compared to thin-film [27]. Moreover, the vertical nanostructures have beneficial effects for DSSCs, since they have antireflection properties through the nanostructures that efficiently trap more light. Therefore, this method can be employed for developing high surface area photoanode nanostructures for DSSC applications and other optoelectronic applications.

### 3.2. TEM Analysis

For TEM analysis, the TiO<sub>2</sub> nanostructure layer deposited on the glass substrate was scrapped out using a doctor blade, which dispersed the scrapped-out powder into the acetone in a vial and ultrasonically sonicated the sample properly for a few minutes for good dispersion. Finally, place a drop of sonicated solution onto the TEM grid for TEM analysis. The TEM analysis of the TiO<sub>2</sub> and TAT samples is shown in Fig. 3 (a) and (b). The TiO<sub>2</sub>-NWs are successfully grown using the GLAD technique, as shown in Fig. 3(a). The typical length measured from the nanowire image is  $\sim 259$  nm and the arrow mark indicates the growth direction. Further, the TEM image of TAT sample manifests the presence of Ag-NP assisted at the mid of the TiO<sub>2</sub>-NW. The HR-TEM images in inset (1) and (2) of Fig. 3(b) show the

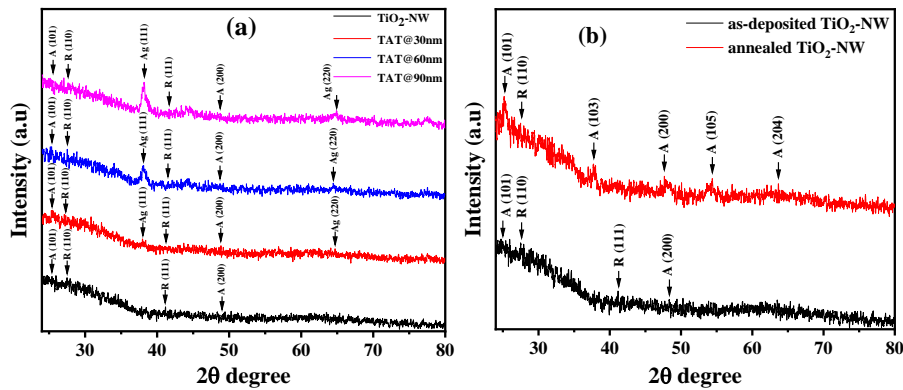
presence of Ag crystal and TiO<sub>2</sub> crystal. The measured lattice constant from the inset (1) is found to be  $\sim 0.35$  nm, which corresponds to (101) crystal plan of anatase TiO<sub>2</sub> (JCPDS No. 75-1753). From the inset (2), the measured lattice constant is found to be  $\sim 0.24$  nm in the related crystal plane (111) of the Ag crystal (JCPDS No. 04-0783).



**Fig. 3** (a) The TEM image of the as deposited TiO<sub>2</sub>-NW, b) TEM image of annealed TAT@30nm and the inset represent the magnified HR-TEM image

### 3.2.1 XRD analysis

The as-deposited TiO<sub>2</sub>-NW and Ag-NP (30 nm, 60 nm and 90 nm) assisted vertically aligned TiO<sub>2</sub>-NW samples are analyzed by X-ray diffraction (XRD). Fig. 3(a) shows the XRD results of the as-deposited TiO<sub>2</sub>-NW, TAT@30nm, TAT@60nm, and TAT@90nm samples. The weak peaks observed at  $2\theta = 25.76^\circ$ ,  $37.58^\circ$ ,  $48.4^\circ$  and  $63.4^\circ$  are attributed to TiO<sub>2</sub> crystals with the corresponding orientation of (101), (103), (200) and (204), respectively (JCPDS No. 75-1753). The weak peaks may correspond to the small grain size of TiO<sub>2</sub> crystal grains. Again, the peaks at  $38.27^\circ$ ,  $34.72^\circ$  and  $77.33^\circ$  are related to the (111), (220) and (310) planes of Ag crystals (JCPDS No. 04-0783).

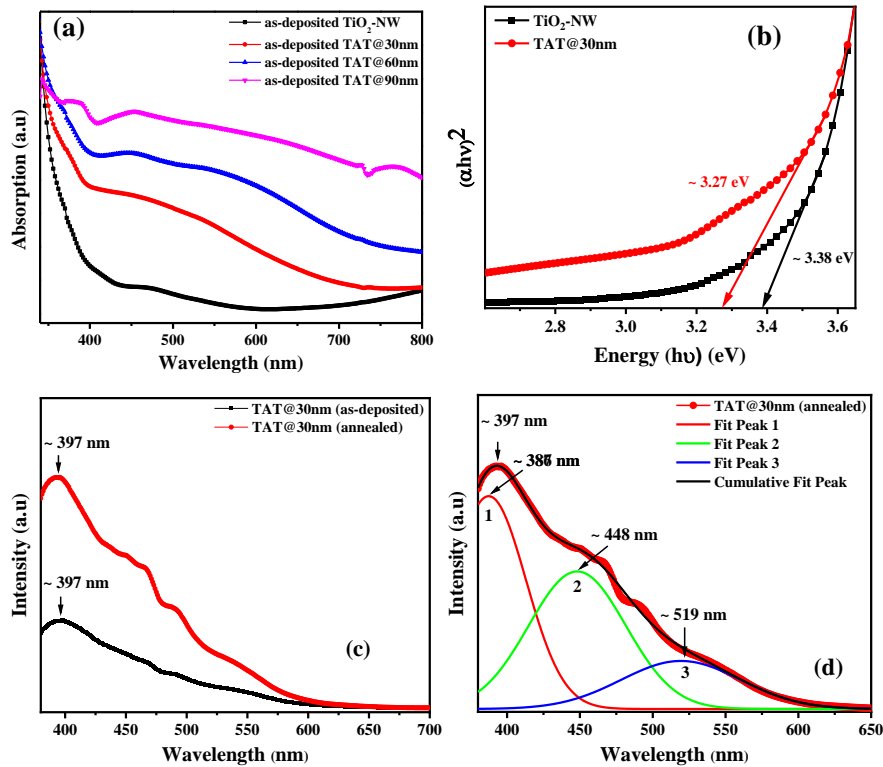


**Fig. 4** (a) Shows the XRD results of TiO<sub>2</sub>NW, TAT@30nm, TAT@60nm, and TAT@90nm samples deposited at room temperature and (b) shows the XRD peak results for as-deposited TiO<sub>2</sub>-NW and annealed TiO<sub>2</sub>-NW

The annealing of TiO<sub>2</sub>-NW improves the crystalline structure of TiO<sub>2</sub>, as shown in Fig. 4(b). Oblique deposition of titanium oxide layers for DSSCs is done by using reactive e-beam deposition, which has the same weak peak. Further, as-deposited TiO<sub>2</sub> film was annealed for 3 h at 500 °C to produce crystalline TiO<sub>2</sub> [28].

### 3.3. UV-Vis Spectroscopy and Photoluminescence Spectroscopy

The optical properties of TiO<sub>2</sub>-NW and TAT specimens fabricated on an FTO substrate were analyzed in the wavelength range of 340 nm to 800 nm using a UV-Vis spectrophotometer. The recorded absorption intensity of the sample is shown in Fig. 5 (a). The absorption spectrum of TiO<sub>2</sub>-NW shows a higher absorption peak in the ultraviolet range. This peak may be attributed to electron excitation from the outermost valence band (VB) to the conduction band (CB) of the TiO<sub>2</sub> [29]. Moreover, the absorption spectrum of the TiO<sub>2</sub>-NW is significantly enhanced in the visible region after the incorporation of different NP sizes, i.e., 30 nm, 60 nm and 90 nm. This significant improvement at around 400 to 600 nm in the absorption spectrum may be due to the SPR effect of Ag-NP [30]. Moreover, the



**Fig. 5** (a) the optical absorption spectra of as-deposited TiO<sub>2</sub>-NW, TAT@30nm, TAT@60nm and TAT@90nm specimens fabricated on the FTO substrate, (b)-the band gap of as-deposited TiO<sub>2</sub>-NW and TAT@30nm, (c) PL spectra of as-deposited and annealed TAT@30nm samples and (d) Shows the Gaussian fitted PL graph of annealed TAT@30nm sample

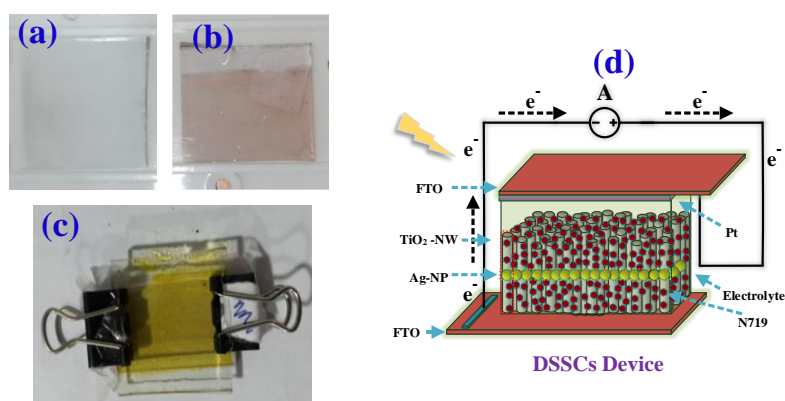


calculated band gaps from the tauc plot of TiO<sub>2</sub>-NW and TAT@30nm are  $\sim 3.38$  and  $\sim 3.27$  eV, respectively. Further, room temperature photoluminescence (PL) analysis of the Ag-NP assisted TiO<sub>2</sub>-NW sample was done at an excitation wavelength of 340 nm by using a 370 nm stopband filter. The broad PL intensity of the as-deposited and annealed TAT@30nm samples is plotted in Fig. 5 (b). A broad emission peak at  $\sim 397$  nm was observed from as-deposited and annealed TAT@30nm samples.

It was also observed that the PL intensity of the annealed TAT@30nm specimen increased compared to the as-deposited sample, which may be due to the increase in the crystallinity of TiO<sub>2</sub> by reducing the oxygen vacancies. Further, the gaussian fitted curve shows the peaks at  $\sim 385$  nm,  $\sim 448$  nm, and 519 nm, which correspond to the band-to-band transition of TiO<sub>2</sub> and oxygen defects present between the band gap, as shown in Fig. 5(d).

### 3.4. Device characterization

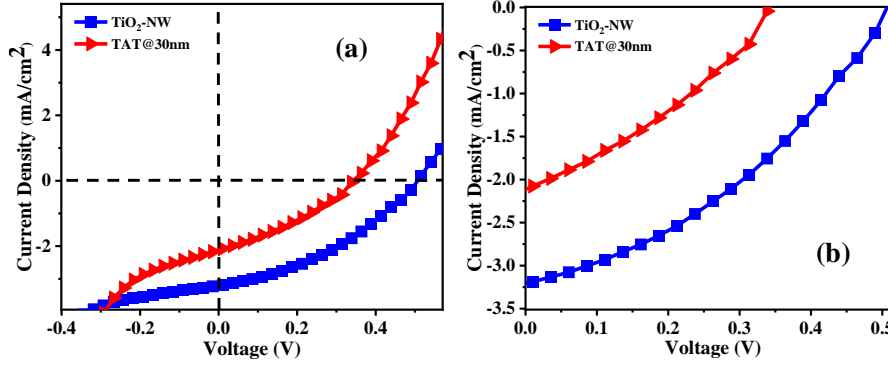
The electrical performance of the fabricated DSSCs is characterized at room temperature by using a Source Meter (Keithley 2450) connected to the computer and the photocurrent measurement was taken under light illumination at 100 mW/cm<sup>2</sup> powered by a solar simulator (SS150, Scientech, Canada). Fig. 6 shows the dye absorbed photoanode, counter electrode and DSSC device. The schematic of the DSSC device based on the TAT photoanode is shown in Fig. 6(d). Fig. 7 shows the *J-V* graph of DSSCs based on TiO<sub>2</sub>-NW and TAT@30nm photoanodes. And, Table I shows the cell performance of DSSCs devices and the corresponding photovoltaic parameters.



**Fig. 6** (a) Fabricated counter electrode, (b) Dye absorbed photoanode and (c) Fabricated DSSC device based on TiO<sub>2</sub>-NW photoanode and (d) Schematic of DSSC device based on TAT photoanode

The PCE of the TiO<sub>2</sub>-NW is  $\sim 0.61\%$  and the corresponding open-circuit voltage ( $V_{oc}$ ) and short circuit current density ( $J_{sc}$ ) of the cell are  $\sim 0.51$  V and  $\sim 3.21$  mA/cm<sup>2</sup>. The efficiency of the DSSCs is reduced to  $\sim 0.24$  percent after the incorporation of Ag nanoparticles, with the corresponding  $V_{oc}$  and  $J_{sc}$  being  $\sim 0.34$  V and  $\sim 2.11$  mA/cm<sup>2</sup>, respectively. So, there is a difference between the  $J_{sc}$  that depends on the light conversion activity and the structure of the photoanode, which determines the electron diffusion PCE

of the solar cell. It is observed that TiO<sub>2</sub>-NW photoanode based DSSC devices show better efficiency compared to TAT@30nm photoanode based devices.



**Fig. 7** a)  $J$ - $V$  graphs of DSSCs based on TiO<sub>2</sub>-NW and TAT@30nm photoanode, b) magnified  $J$ - $V$  graphs of DSSCs

**Table 1** Photovoltaic performance of TiO<sub>2</sub>-NW and TAT@30nm photoanode based DSSCs

Photoanode	$V_{oc}$ (V)	$J_{sc}$ (mA)/cm <sup>2</sup>	$V_m$ (V)	$I_m$ (mA)	$FF$	$\eta$ %	Reference
MWCNT	0.28	1.76	--	--	0.30	0.15	[31]
ZnO film	0.49	2.15	--	--	0.54	0.56	[32]
TiO <sub>2</sub> film	0.56	1.17	--	--	0.85	0.56	[33]
TiO <sub>2</sub> -NW	0.51	3.21	0.32	1.94	0.37	0.61	Our result
TAT@Ag 30 nm	0.34	2.11	0.21	1.13	0.34	0.24	Our result

However, the TiO<sub>2</sub>-NW based device improves the accessibility of the entire surface to the dye and corresponding electrolyte medium, leading to a direct and shorter path for the transportation of the electrons. Marquesa *et al.* reported an efficiency of ~ 1.2% from the DSSC fabricated using the tape casting method. It was also observed that the highest efficiency was achieved by using 4-tert-butyl pyridine electrolytes [34]. Again, Erande *et al.* reported a PCE of 0.2% in which the TiO<sub>2</sub> film was deposited using a chemical method [35]. The natural dye, acting as a sensitizer of DSSCs, was less efficient. Even so, the efficiency of our DSSC device based on TiO<sub>2</sub>-NW was higher than that of DSSCs using natural dye. Furthermore, our device shows better performance in terms of efficiency compared to some of the recently reported devices, as shown in table I. However, the efficiency of our device is still low, which may be due to the small thickness of the photoanode. So, the efficiency may be further improved by increasing the TiO<sub>2</sub>-NW photoanode thickness and also by reducing the size of the metal nanoparticles.

#### 4. CONCLUSION

In conclusion, the GLAD method was used to develop TiO<sub>2</sub>-NW and Ag-NP-assisted TiO<sub>2</sub>-NW photoanodes on an FTO substrate for the development of DSSCs. The SEM and TEM analysis reveal the successful deposition of TiO<sub>2</sub>-NW and TAT nanowires. The XRD investigation reveals the presence of Ag-NP and TiO<sub>2</sub> crystals in the samples. The

absorption enhancement from the Ag-NP assisted TiO<sub>2</sub>-NW samples observed in the absorption spectrum may be due to the SPR effect of Ag-NP present in the TiO<sub>2</sub>-NW. The TiO<sub>2</sub>-NW based DSSC device shows better efficiency compared to the Ag-NP assisted TiO<sub>2</sub>-NW photoanode based DSSC device. It may be concluded that the size of the Ag-NP incorporation at the mid-point of the TiO<sub>2</sub> NW needs to be reduced to enhance the efficiency of DSSC. Therefore, this presented technique may be employed for developing DSSCs and other optoelectronic device applications.

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