J. Electrochem. Sci. Eng. 10(4) (2020) 347-360; http://dx.doi.org/10.5599/jese.806



Open Access: ISSN 1847-9286 www.jESE-online.org

Original scientific paper

Study of oxygen evolution reaction on thermally prepared $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes

Ollo Kambiré^{1, \boxtimes}, Lemeyonouin A. G. Pohan², Konan H. Kondro³ and Lassiné Ouattara^{3, \blacksquare}

¹UFR Sciences et Technologies, Université de Man, BP 20 Man, Côte d'Ivoire ²UFR Sciences Biologiques, Université Peleforo Gon Coulibaly de Korhogo, BP 1328 Korhogo, Côte d'Ivoire

³Laboratoire de constitution et réaction de la matière, UFR SSMT, Université Félix Houphouët-Boigny de Cocody, Abidjan, 22 BP 582 Abidjan 22, Côte d'Ivoire

Corresponding authors: [⊠]<u>kambireollo@yahoo.fr</u>; [⋸]<u>ouatlassine@yahoo.fr</u>

Received: March 9, 2020; Revised: June 21, 2020; Accepted: July 10, 2020

Abstract

The mixed coupled $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes (x = 0, 10, 20, 30, 40, 50, 60, 70, 80, 90 and 100) were thermally prepared at 450 °C on titanium supports. The prepared electrodes were firstly physically characterized by scanning electron microscopy (SEM), X-ray diffraction (XRD) and X-ray photoelectron spectroscopy (XPS). Afterwards, electrochemical characterizations were performed by voltammetric (cyclic and linear) methods in different electrolyte media (KOH and $HClO_4$). It is shown that the prepared electrodes are composed by both PtO_y (platinum and platinum oxide) and IrO₂ (iridium dioxide). For _xPtO_{y⁻(100-x)}IrO₂ electrodes having higher content of IrO₂, more surface cracks and pores are formed, defining a higher surface area with more active sites. Higher surface area due to presence of both PtO_v and IrO₂, is for $_{x}$ PtO_v-(100 x_{i} lrO₂ electrodes in 1 M KOH solution confirmed by cyclic voltammetry at potentials of the oxide layer region. For all prepared electrodes, voltammetric charges were found higher than for PtO_v, while the highest voltammetric charge is observed for the mixed $_{40}$ PtO_v- $_{60}$ IrO₂ (x = 40) electrode. The Tafel slopes for oxygen evolution reaction (OER) in either basic (0.1 M KOH) or acid (0.1 M HClO₄) media were determined from measured linear voltammograms corrected for the ohmic drop. The values of Tafel slopes for OER at $PtO_{y, 90}PtO_{y-10}IrO_2$ and IrO_2 in basic medium are 122, 55 and 40 mV dec⁻¹, respectively. For other mixed electrodes, Tafel slopes of 40 mV dec⁻¹ were obtained. Although proceeding by different OER mechanism, similar values of Tafel slopes were obtained in acid medium, i.e., Tafel slopes of 120, 60 and 39 mV dec⁻¹ were obtained for PtO_{y} , $_{90}PtO_{y^{-10}}IrO_2$ and IrO_2 , and 40 mV dec⁻¹ for other mixed electrodes. The analysis of Tafel slope values showed that OER is more rapid on coupled mixed electrodes than on pure PtO_{y} . For mixed $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes, OER is more rapid when the molar percent of PtO_y meets the following condition: $0 < x \le 80$. This study also showed that the mixed coupled electrodes are more electrocatalytically active for OER than either PtO_y or IrO_2 in these two media.

Keywords

Platinum oxide, iridium dioxide, mixed Pt-Ir oxide, oxygen evolution, Tafel slope, electrocatalytic activity

Introduction

Currently, climate changes and energy shortage are some of the greatest challenges for humanity [1-3]. Increasing carbon dioxide (CO₂) levels in the atmosphere cause severe issues such as global warming, energy security, and sustainability. These issues stimulated researchers to find new clean technologies that are high in efficiency and low in cost [4]. To fulfill this goal without destroying the environment, alternative methods of energy generation, conversion, and storage need to be employed [5-8]. As a sustainable and environmentally friendly alternative, water electrolysis combined with renewable electricity sources, such as wind and solar power, is expected to provide a practical solution. However, the major drawback of technologies such as solar or wind energy is that they cannot function continuously. Thus, renewable energy from water splitting by the electrochemical route is considered as one of the important techniques to provide a solution for climate change and energy shortage. The overall water splitting reaction includes the hydrogen evolution reaction (DER). To realize the practical and sustainable application of electrocatalytic hydrogen/oxygen evolution reactions (HER/OER), an efficient, durable, and economical electrocatalyst is demanded triggering the reaction with minimal overpotential and fast kinetics [9-11].

According to the literature, the high-performance electrocatalysts are mainly based on noble metals such as for example, platinum [12,13]. From the cost perspective, however, it is highly desirable to develop electrocatalysts based on metal oxides.

The most usually targeted OER electrocatalyst oxides are iridium oxide (IrO₂) and ruthenium oxide (RuO₂) [14-16]. Ruthenium oxide exhibits several interesting properties, and has the advantages of high conductivity and good electrochemical reversibility [17-19]. However, the ruthenium oxide electrode is not applicable for long-term operations and dissolves during OER. Iridium oxide has attracted considerable attention due to its unique advantages including excellent chemical stability and sensitivity, impressive electrocatalytic activity, and sufficient electric conductivity [20-22]. Besides, the iridium oxide is known for its superb biocompatibility and long-term stability [23-27].

It was already found that when IrO_2 is mixed with Pt oxide, Pt-Ir oxide shows enhanced electrocatalytic activity and rapid oxygen evolution reaction [28,29]. To further research about electrocatalytic properties of mixed PtO_y - IrO_2 , it appears necessary to optimize the ratio of PtO_y and IrO_2 on this electrode. In the present work, the proportion of each element (Pt and Ir) of the mixed electrode has been varied and the influence of these variations on the kinetics of OER is studied. Mixed $_xPtO_{y^-(100-x)}$ IrO_2 electrodes (x = 0, 10, 20, 30, 40, 50, 60, 70, 80, 90 and 100) where x represents the molar percent of PtO_y precursor, have been prepared thermally at 450 °C and the kinetics of OER is followed on these electrodes in 0.1 M KOH and 0.1 M HClO₄ media, respectively.

Experiments

Metallic supports (or substrates) of titanium plates of dimensions $16 \times 16 \times 0.5$ mm were used. These supports were pretreated by sandblasting and rinsing to remove all the impurities and to increase the roughness which improves adhesion of the deposit. The surfaces were degreased and cleaned with isopropanol, followed by rinsing with distilled water. Precursors of H₂PtCl₆×6H₂O (Fluka, 99 %), and H₂IrCl₆×H₂O (Fluka, 99 %) were applied by brushes to the surfaces of the pretreated electrodes. These precursors were diluted in isopropanol (Fluka, 99.5 %). Two precursor solutions of H₂PtCl₆×H₂O and H₂IrCl₆×H₂O were prepared separately at 23.30 g/L and 23.25 g/L, respectively. Then, mixtures of these two solutions were prepared, respecting the proportions by moles number of PtO_y and IrO₂ for the coupled mixed electrodes. The concentrations of precursors in the mixture which served for the preparation of the mixed _xPtO_y-(100-x)IrO₂ electrodes are grouped in Table 1. The solvent of the deposit was evaporated by baking in the oven at 80 °C for 10 min. Then the deposit underwent thermolysis in the furnace for 15 min at 450 °C. Several layers of deposit were thus produced before being calcined in the furnace at 450 °C for one hour. The mixed coupled _xPtO_y-(100-x)IrO₂ electrodes (x = 0, 10, 20, 30, 40, 50, 60, 70, 80, 90 or 100) were prepared by this method, where x is referred to the molar percent of PtO_y precursor. For the mixed electrodes, a mixture of different percent molars of the corresponding precursors was used. The deposit loading was about 5 g m⁻² on each titanium substrate.

Electrodes	Concentration of H ₂ PtCl ₆ ×H ₂ O, M	Concentration of H2IrCl6×H2O, M
10PtOy-90IrO2	0.0045	0.0405
₂₀ PtO _y - ₈₀ IrO ₂	0.0090	0.0360
30PtOy-70IrO2	0.0135	0.0315
$_{40}$ PtO _y - $_{60}$ IrO ₂	0.0180	0.0270
50PtOy-50IrO2	0.0225	0.0225
₆₀ PtO _y - ₄₀ IrO ₂	0.0270	0.0180
₇₀ PtO _y - ₃₀ IrO ₂	0.0315	0.0135
₈₀ PtO _y -20IrO ₂	0.0360	0.0090
₉₀ PtO _y -10IrO ₂	0.0405	0.0045

Table 1. Concentrations of $H_2PtCl_6 \times H_2O$ and $H_2IrCl_6 \times H_2O$ in the mixed coupled precursors used forpreparation of mixed electrodes

High-resolution images of the prepared electrode surfaces were realized by scanning electron microscopy (SEM, ZEISS, SUPRA 40VP). The crystalline structure was examined by X-ray diffraction (XRD) using a Siemens equipment with a Cu cathode. The XPS analysis was carried out with a Kratos Axis-Ultra Spectrometer using a monochromatic Al Kα X-Ray source, operated at 15 kV, and a pass energy of 20 eV.

The electrochemical measurements (cyclic and linear sweep voltammetry) were carried out using a conventional three-electrode electrochemical cell and Autolab PGStat 20 (Ecochemie). The prepared $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes, having geometric surface area of 0.785 cm², served as the working electrodes (WE). The reference electrode (RE) was a saturated calomel electrode (SCE), mounted in a Luggin capillary and placed close to WE by a distance of 1 mm. The counter-electrode (CE) was the spiral platinum wire, having the area at least hundred times larger than WE. All potential values in this manuscript are given against the reversible hydrogen electrode (RHE) using the following relation:

 $E_{\rm RHE} = E_{\rm SCE} + 0.059 \, \rm pH + 0.244 \, \rm V$

All electrochemical experiments were made at the ambient temperature of 25 °C.

(1)

Results and discussion

Physical characterization of prepared electrodes

The prepared electrodes were analyzed by scanning electron microscopy. The surface micrographs of IrO₂, $_{40}$ PtO_y- $_{60}$ IrO₂, $_{70}$ PtO_y- $_{30}$ IrO₂, and PtO_y electrodes are presented in Figure 1. The micrograph of pure IrO₂ in Figure 1a shows superimposed layers of iridium dioxide, cracks and some pores, what is in accordance with the literature [24]. The cracks observed on the surface of pure IrO₂ occurred during the cooling of the deposit due to the mechanical stresses produced by the difference in the coefficients of thermal expansion of the substrate and the deposit. In Figure 1b, SEM image of the electrode containing $_{40}$ PtO_y- $_{60}$ IrO₂ shows that the surface of this deposit appears more continuous and the cracks are narrower in comparison with pure IrO₂. Also, some small Pt crystals dispersed on the surface of the deposit can be observed in Figure 1b. SEM image of the deposit containing $_{70}$ PtO_y- $_{30}$ IrO₂ is presented in Figure 1c, showing a porous and very rough surface with almost non-existent cracks. At this stage, the layer of the deposit resembles that of PtO_y [25]. On PtO_y electrode surface shown in Figure 1d, there are no cracks but the grains are strongly bonded to each other, thus showing the porous and rough nature of the deposit encreases, the surface structure of the deposit tends to be more like that of PtO_y.



Figure 1. Scanning electron micrographs of IrO_2 (a), $_{40}PtO_{y-60}IrO_2$ (b), $_{70}PtO_{y-30}IrO_2$ (c) and PtO_y (d)

XPS spectra of $PtO_{y, 50}PtO_{y-50}IrO_2$ and IrO_2 electrodes are presented in Figure 2. On the XPS spectrum of PtO_y , one can notice the presence of three intense peaks at 70, 312 and 335 eV, which all characterize platinum presence on titanium. Carbon is also present with a peak observed at 286 eV at all presented electrodes. Presence of carbon could be due to the poisoning of the measuring device, isopropanol used in the precursor preparations, and/or contamination in the oven used to prepare the electrodes. Two very close peaks are observed at 218 and 224 eV. These two peaks characterize the presence of platinum oxide. In the XPS spectrum of the mixed electrode

 $_{50}$ PtO_y- $_{50}$ IrO₂ in addition to platinum peaks observed at 70, 312 and 335 eV, there are some new peaks at 61, 296 and 494 eV. These three new peaks also exist in XPS spectrum of pure iridium oxide, approving the presence of iridium oxide in the mixed electrode.



Figure 2. XPS spectra of PtO_y, ₅₀PtO_y-₅₀IrO₂ and IrO₂.

The XRD spectrum of PtO_y presented in Figure 3 provides more details on the bulk structure of the platinum coating. The XRD pattern of PtO_y shows two peaks at $2\theta = 40$ and 68° , which are assigned to the metallic Pt (JCPDS N° 4-802). The peaks located at 35 and 63° correspond to PtO₂ (JCPDS N° 38-1355). The peaks located at $2\theta = 40$ and 53° could be assigned to Ti [29]. Those XRD results indicate that the coating of PtO_y is a mixture of metallic Pt and PtO₂. In other words, the deposit is a mixture of Pt with zero-valence and its oxidized form, what also confirms the XPS results of PtO_y. Generally, both XPS and XRD results reveal that platinum present at the titanium surface exists in different oxidized states.



Figure 3. XRD spectrum of PtOy

Electrochemical characterization of prepared electrodes with ferri/ferrocyanide redox couple

Figure 4a shows the cyclic voltammograms of the mixed $_{60}PtO_{y}-_{40}IrO_2$ electrode in 1 M KOH, containing different concentrations of potassium ferri-ferrocyanide couple (1 - 100 mM). These voltammograms showed oxidation peaks in the forward potential scans and reduction peaks in the

reverse potential scans. There is an increase in the current density of both peaks with increase of the concentration of potassium ferri-ferrocyanide couple. The current densities of the oxidation and reduction peaks were plotted as a function of the concentration of potassium ferri-ferrocyanide for ${}_{10}PtO_{y-90}IrO_2$, ${}_{50}PtO_{y-50}IrO_2$, and ${}_{90}PtO_{y-10}IrO_2$ electrodes, and presented in Figures 4b and 4c. The obtained curves are straight lines that pass through the origin of the coordinate system. Note that similar results were obtained with other mixed electrodes studied here (curves not shown). The equations of the lines make it possible to determine the anode (D_a) and cathode (D_c) diffusion coefficients, using the following equation [30,31]:

 $i_{\rm p}$ = 2.69 10⁵ n^{3/2}AD^{1/2}Cv^{1/2}

In eq. (2), j_P is peak current density, n is number of exchanged electrons, A is geometric surface area, C is concentration of reaction species, D is their diffusion coefficient, and v is scan rate.



Figure 4: Cyclic voltammetry of mixed ${}_{60}PtO_{y^-40}IrO_2$ electrode in 1 M KOH containing different concentrations of ferri-ferrocyanide redox couple at 20 mV s⁻¹ (a), anodic peak current densities (b) and cathodic peak current densities (c) against concentrations of [Fe(CN)₆]³⁻/[Fe(CN)₆]⁴⁻ at mixed ${}_{10}PtO_{y^-90}IrO_2$, ${}_{50}PtO_{y^-50}IrO_2$ and ${}_{90}PtO_{y^-10}IrO_2$ electrodes.

The values of diffusion coefficients of ferri/ferrocyanide couple listed in Table 2 were determined by eq. (2) with n = 1, and are of the same order of magnitude as those from the literature [33]. The certain difference observed between the values determined in this work and literature data could be linked to the use of the geometric surface and not the real surface of the electrode in the calculation. It is clear from Table 2 that *D* values are somewhat lower for mixed _xPtO_y-(100-x)IrO₂ electrodes than those obtained on pure IrO₂ and PtO_y.

The ratios of the absolute value of the anode and cathode peak current densities are also shown in Table 2. A value close to 1 was obtained for each electrode. These results confirm an almost reversible behavior of the potassium ferri-ferrocyanide couple on all here studied electrodes.

Electrodes	 j Pa / j Pc	D _a / 10 ⁻⁶ cm ² s ⁻¹	D _c / 10 ⁻⁶ cm ² s ⁻¹
PtOy	1.03	9.43	9.77
₉₀ PtO _y -10IrO ₂	1.00	5.19	6.11
80PtOy-20IrO2	1.01	5.24	5.88
70PtOy-30IrO2	0.99	3.27	5.02
60PtOy-40IrO2	1.02	3.62	5.04
50PtOy-50IrO2	1.04	4.57	6.33
40PtOy-60IrO2	1.00	2.70	4.60
30PtOy-70IrO2	0.99	2.50	4.29
20PtOy-80IrO2	1.01	4.47	5.20
10PtOy-90IrO2	1.03	4.37	3.65
IrO ₂	1.06	12.29	12.92

Table 2: Kinetic parameters of ferri/ferrocyanide redox reaction in 1 M KOH at xPtOy-(100-x)IrO2 electrodes

Electrochemical characterization of prepared electrodes in KOH medium

Figure 5a shows cyclic voltammograms of PtO_y and IrO₂ in the basic medium (0.1 M KOH) in the oxide stability potential zone (0.77 to 1.47 V vs. RHE). For pure PtO_v, a presence of some reduction current can be noticed from 0.77 to 0.91 V vs. RHE. This can be ascribed to the reduction of oxides formed in the forward potential scan [33] and also during the electrode calcination. A rectangular shape from 0.91 to 1.47 V/RHE is characterized by an almost constant current density and found similar to that observed previously [29]. In comparison to IrO₂ electrode, PtO_v electrode has much lower voltammetric charge. At the other side, IrO₂ electrode possesses very high voltammetric charge and its voltammetric response is generally characteristic for heat-prepared DSAs (dimensionally stable anodes) [29,35,36]. Much improved voltammetric charge of the pure IrO_2 electrode can be associated with the pseudocapacitive reactions at IrO2 and also, due to the presence of cracks and pores that increased the number of electrode active sites at the surface [35]. Figure 5b shows cyclic voltammograms of mixed _xPtO_y-(100-x)IrO₂ electrodes in 0.1 M KOH medium from 0.77 to 1.47 V/RHE. The voltammograms of mixed electrodes have shapes similar to PtO_v, indicating that PtO_v is present on all the prepared mixed electrodes. However, the voltammetric charges of these electrodes are higher than that of PtO_v. This increase in the voltammetric charge is due to the presence of IrO_2 on the surface of mixed electrodes. At mole percentages of PtO_y greater than 50 %, the voltammetric charge is lower than that of IrO₂ but still greater than that of PtO_v. For mole percentages of PtO_v less than 50 %, the voltammetric charge is higher than that of PtO_v and IrO₂. The highest voltammetric charge is observed for PtO_v with mol percentage of 40 %. The high voltammetric charge of mixed electrodes shows the increase in active sites due to the presence of both PtO_v and IrO₂.



Figure 5. Cyclic voltammograms (10 mV s⁻¹) in 0.1 M KOH of PtO_y and IrO₂ electrodes (a) and $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes (b).

The studies were also carried out at the lower scan rate because, at a low scan rate, the system is in a quasi-stationary state. In Figure 6, the linear voltammograms obtained with different _xPtO_y-(100-x)IrO₂ electrodes in the basic medium of 0.1 M KOH are presented. These voltammograms show that the potential at j = 1 mA cm⁻² related to oxygen evolution varies from one electrode to another. The OER potential was also determined at j = 10 mA cm⁻² for each electrode. The potential values of all prepared _xPtO_y-(100-x)IrO₂ electrodes at these two current density values are presented in Table 3.



Figure 6. Linear voltammograms (5 mV s⁻¹) of prepared electrodes in 0.1 M KOH medium

Table 3. Potentials of oxygen evolution reaction on various ${}_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes in 0.1 M KOH at $j = 1 \text{ mA cm}^{-2}$ and $j = 10 \text{ mA cm}^{-2}$, scan rate = 5 mV s⁻¹.

Electrode	<i>E</i> / V <i>vs.</i> RHE ($j = 1 \text{ mA cm}^{-2}$)	<i>E</i> / V <i>vs.</i> RHE (<i>j</i> = 10 mA cm ⁻²)
PtOy	1.679	1.908
90PtOy-10IrO2	1.521	1.710
80PtOy-20IrO2	1.488	1.578
70PtOy-30IrO2	1.473	1.571
60PtOy-40IrO2	1.470	1.576
50PtOy-50IrO2	1.468	1.568
40PtOy-60IrO2	1.460	1.544
30PtOy-70IrO2	1.453	1.531
20PtOy-80IrO2	1.464	1.543
10PtOy-90lrO2	1.467	1.557
IrO ₂	1.489	1.576

Data in Table 3 show that OER potentials for ${}_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes determined at j = 1 mA cm⁻² and j = 10 mA cm⁻² vary in a similar way. The highest potential values observed at any current density for PtO_y electrode suggest that it is less electrocatalytically active for OER than other electrodes in the basic medium. Also, it seems that ${}_{90}PtO_{y^{-}10}IrO_{2}$ electrode is more electrocatalytically active than PtO_y, but less than IrO₂ and other mixed electrodes. On the other hand, the electrodes with PtO_y percentage lower or equal to 80 % have a potential related to OER close but less than pure IrO₂. It is clear that the coupling of PtO_y with IrO₂ has improved electrocatalytic activity of the electrode. The electrodes which have PtO_y percentage lower or equal to 80 % are the most electrocatalytically active for the oxygen evolution in the basic medium.

The Tafel slope of OER on an electrode can be abnormally high, especially within a high potential domain. According to Kapalka *et al.* [37], the increase of the Tafel slope is due to the ohmic drop.

The ohmic drop between the anode and the cathode is due to the geometry of the cell, solution conductivity, presence of bubbles in solution, and accumulation of gas bubbles produced on the surface of the electrode (blockage of the surface) [38,39]. Here, the ohmic drop corrections of the measured polarization curves were carried out according to the method already described in the literature [28,29]. Briefly speaking, the overpotential (η / V) observed in the experiment can be defined by

$$\eta = a + b \ln j + jR \tag{3}$$

where a / V is Tafel constant, $b / V dec^{-1}$ is Tafel slope, $j / A cm^{-2}$ is current density and $R / \Omega cm^{-2}$ is the total area-specific, uncompensated resistance of the system, assumed to be constant. The derivative of eq. (3) concerning current density, results in eq. (4) from which b and R can be easily obtained by plotting $d\eta / dj$ as a function of 1 / j.

$$d\eta/dj = (b/j) + R \tag{4}$$

The estimation of R allows correcting the experimental overpotential by subtracting the ohmic drop jR according to the following equation:

$$\eta_{\rm corr} = \eta - jR \tag{5}$$

where $\eta_{\rm corr}$ stands for the corrected overpotential.

During the calculations, the derivatives $d\eta / dj$ were replaced by their finite elements $\Delta \eta / \Delta j$ estimated from each pair of consecutive experimental points.

After correcting the ohmic drop for each electrode in the electrolyte used, the Tafel slope lines were determined from the corrected *E vs.* log *j* dependences that are for some $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes presented in Figure 7a.



Figure 7. Tafel lines (a) and Tafel slopes (b) of different $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes in 0.1 M KOH.

Note that Tafel slopes vary from electrode to electrode, suggesting that the kinetics of OER varies from one electrode to another Thus, it can be seen from Figure 7a that the Tafel slope of pure IrO_2 electrode is 40 mV dec⁻¹. According to Figure 7b, Tafel slope remains almost constant at 40 mV dec⁻¹ for all mixed electrodes which have PtO_y percentage lower or equal to 80 %. This result indicates that OER at IrO_2 and all these mixed electrodes proceeds practically with the same kinetics. At the other side, Tafel slopes of 55 and 122 mV dec⁻¹ were respectively obtained for $_{90}PtO_{y-10}IrO_2$ and PtO_y electrodes. These results suggest that the electrodes which have the voltammetric response close to PtO_y have also high Tafel slope similar to PtO_y. This is the case for electrodes whose surfaces are rich in PtO_y. Also, when the surface is dominated by IrO_2 , the Tafel slopes tend towards that of pure IrO_2 .

Generally, the values of the obtained Tafel slopes suggest that the kinetics of OER on mixed electrodes is faster on the electrodes having the percentage of PtO_y lower or equal to 80 % than on the electrodes with the percentage of PtO_y higher than 80 %.

The OER mechanism in the basic media is taken from the literature on mixed PtO_y and IrO_2 electrodes [32], and described by following four steps

$$S + OH^{-} \rightarrow SOH + e^{-}$$

$$SOH + OH^{-} \rightarrow SO^{-} + H_{2}O$$

$$SO^{-} \rightarrow SO + e^{-}$$

$$(6)$$

$$(7)$$

$$(7)$$

$$(8)$$

$$2SO \rightarrow 2S + O_2 \tag{9}$$

where S stands for electrode active sites, while OH, O⁻ and O represent adsorbed intermediates.

The mechanism described by eqs. (6-9) predicts the Tafel slope of 120 mV dec⁻¹ when step described by eq. (6) is the rate determining step (rds), 60 mV dec⁻¹ when step (7) is rds, 40 mV dec⁻¹ when step (8) is rds, and 15 mV dec⁻¹ when step described by eq. (9) is rds.

Tafel slopes presented in Figure 7b suggest that mixed $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes have probably the same reaction pathway for OER, but different rate determining steps. For PtO_y anode, the first step described by eq. (6) can be considered as the rate determining step, because the observed Tafel slope is close to 120 mV dec⁻¹. On the pure IrO₂ and the anodes with PtO_y percentage less or equal to 80 %, the observed Tafel slope is 40 mV dec⁻¹, indicating that the step described by eq. (8) could be considered as the rate determining step. For the electrode containing 90 % of PtO_y, the second step described by eq. (7) will be considered as the rate determining step, because its Tafel slope was found close to 60 mV dec⁻¹.

Electrochemical characterization of prepared electrodes in perchloric acid medium

The oxygen evolution has also been studied in an acid medium. The linear voltammograms of mixed $_xPtO_{y^{-}(100-x)}IrO_2$ electrodes in 0.1 M HClO₄, recorded in a quasi-stationary manner are presented in Figure 8.



Figure 8. Linear voltammetry (5 mV s⁻¹) of $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes in 0.1 M HClO₄

Figure 8 shows that the onset potential related to the oxygen evolution (the potential of the sharp current increase) varies from one electrode to another but is the highest for PtO_y electrode. The potential values at j = 1 mA cm⁻² and j = 10 mA cm⁻² related to OER were determined for all $_xPtO_{y^{-}(100-x)}IrO_2$ electrodes and the obtained values are represented in Table 4.

Data in Table 4 show that potential values of OER vary from one electrode to another similarly and regardless of the current density. PtO_y electrode has the highest potential related to oxygen evolution, suggesting that PtO_y is the least electrocatalytically active electrode for OER in an acid medium. Except for the electrode which contains 90 % PtO_y , the mixed $_xPtO_{y^-(100-x)}IrO_2$ electrodes have lower onset potentials related to OER than either pure PtO_y or pure IrO_2 . This pointed out that the mixed Pt-Ir oxide electrode in an acid medium is generally more electrocatalytically active for O_2 evolution than its pure components. These results are fully consistent with those observed for OER in the basic medium.

Electrode	<i>E /</i> V <i>vs.</i> RHE (<i>j</i> = 1 mA cm ⁻²)	<i>E</i> / V <i>vs.</i> RHE (<i>j</i> = 10 mA cm ⁻²)
PtOy	1.571	1.714
90PtOy-10IrO2	1.470	1.587
80PtOy-20IrO2	1.434	1.479
70 PtO y-30 lrO 2	1.428	1.475
60PtOy-40IrO2	1.430	1.475
50PtOy-50IrO2	1.438	1.475
40PtOy-60IrO2	1.431	1.458
₃₀ PtO _y - ₇₀ IrO ₂	1.424	1.454
20PtOy-80IrO2	1.429	1.461
10PtOy-90IrO2	1.441	1.481
IrO ₂	1.450	1.514

Table 4. Potentials of oxygen evolution reaction on various $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes in 0.1 M HClO₄ at j = 1 mA cm⁻² and j = 10 mA cm⁻² and scan rate = 5 mV s⁻¹

The Tafel slopes for OER in 0.1 M HClO₄ were determined for all studied $_xPtO_{y^-(100-x)}IrO_2$ electrodes. As shown in Figure 9a, plotting of the corrected *E vs*. log *j* gave straight lines in the Tafel region. The particular values of Tafel slopes for all studied $_xPtO_{y^-(100-x)}IrO_2$ electrodes are recorded in Figure 9b.



Figure 9. Tafel lines (a) and Tafel slopes (b) of different $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes in 0.1 M HClO₄ at 5 mV s⁻¹

Tafel slope of 120 mV dec⁻¹ was obtained for pure PtO_y, while 60 mV dec⁻¹ was obtained for the mixed electrode with 90 % of PtO_y. The other mixed electrodes and pure IrO₂ gave values of 40 mV dec⁻¹. All these values obtained in the acid medium (0.1 M HClO₄) are almost identical to those obtained in the basic medium (0.1 M KOH). Also, the Tafel slopes show that the oxygen evolution reaction in the acidic and basic medium is faster on the electrodes which have a percentage in PtO_y lower or equal to 80 % than on the electrodes which have a percentage in PtO_y higher than 80 %.

The obtained Tafel slopes, together with the literature results [27,29,40] point towards the following mechanism of OER on the prepared electrodes in an acid medium:

$S + H_2O \rightarrow S-OH + H^+ + e^-$	(10)
$S-OH \rightarrow SO + H^+ + e^-$	(11)
$2SO \rightarrow 2S + O_2$	(12)

where S stands for electrode active sites, while OH and O represent adsorbed intermediates.

The mechanism described by eqs. (10-12) predicts the Tafel slope of 120 mV dec⁻¹ if the first step described by eq. (10) is the rate determining step (rds), 40 mV dec⁻¹ if the second step described by eq. (11) is rds, and 30 mV dec⁻¹ if the third step described by eq. (12) is rds.

For OER on PtO_y in HClO₄ medium, the rate determining step is the first step described by eq. (10), because the Tafel slope is 120 mV dec⁻¹. For the pure IrO₂ electrode and the mixed $_x$ PtO_y-(100-x)IrO₂ electrodes except that with 90 % PtO_y, the rate determining step is the second step described by eq. (11), because their Tafel slopes are very close to 40 mV dec⁻¹. With the platinumiridium dioxide coupling with a proportion of 90 % PtO_y, however, the Tafel slope is 60 mV dec⁻¹. This specific value is between that obtained for pure PtO_y (120 mV dec⁻¹) and IrO₂ (39 mV dec⁻¹), but closer to IrO₂ than PtO_y. This result is in agreement with the voltammogram for $_{90}$ PtO_y-10IrO₂ electrode that is closer to IrO₂ than PtO_y.

Generally, Tafel slopes of mixed $_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes are close to that of IrO_{2} and it is just IrO_{2} which determines the oxygen evolution rate. The rate determining step is similar to that obtained with IrO_{2} electrode, but OER is slower on $_{90}PtO_{y^{-}10}IrO_{2}$ anode than IrO_{2} anode. The possible reason could be the mixed kinetics due to the adsorption of unstable species such as S-OH* (equations 13a and 13b), or the synergistic effect of PtO_{y} and IrO_{2} [22]. The mechanism of OER involving eq. (13) instead eq. (10) accounts for possible surface chemical rearrangements. Therefore, the elemental step described by eq. (13b) probably controls the overall rate of OER [41,42].

$$S + H_2O \rightarrow S - OH^* + H^+ + e^-$$
(13a)

$$S-OH^* \rightarrow S-OH$$
 (13b)

In each specific solution, PtO_y , IrO_2 and mixed coupled $_xPtO_{y^-(100-x)}IrO_2$ electrodes studied here, generally follow the same oxygen evolution reaction pathway described by equations (6-9) or (10-13), but with different rate determining steps. Note that OER reaction pathways described by equations (6-9) and equations (10-12) are generally known as the oxide decomposition path and the electrochemical oxide path, respectively [43].

The electrodes with molar percentage of PtO_y lower or equal to 80 % have the lowest Tafel slope values. This shows that the kinetics of OER is faster for these electrodes than for the electrodes which have PtO_y percentage greater than 80 %. This result is in agreement with the voltammetric curves where the onset potential related to oxygen evolution reaction on these electrodes is lower than that obtained with the electrodes with PtO_y content greater than 80 % mol. It is clear from these results that the electrodes which have a voltammetric behavior close to that of iridium oxide have faster OER kinetics than the electrodes which have the PtO_y content greater than 80 % mol. When the mixed oxide surface is dominated by certain metal oxide, the slope of Tafel tends towards that oxide. These observations agree with the scanning electron micrographs because on the electrodes having content less or equal to 80 % there are prominent pores and cracks on these electrodes. This has increased the number of active sites, hence the rapid kinetics of oxygen production.

Conclusions

The physical (SEM, XPS and XRD) and electrochemical (cyclic and linear voltametries) characterizations of mixed coupled deposits of $_xPtO_{y^-(100-x)}IrO_2$ with different molar percent of PtO_y precursor (x = 0, 10, 20, 30, 40, 50, 60, 70, 80, 90 and 100) produced thermally on titanium supports, have shown rather different properties.

Presence of Pt in different oxidation states (Pt and PtO₂) and presence of pure IrO₂ is indicated for mixed coupled $_xPtO_{y^-(100-x)}IrO_2$ deposits, while their surface morphology was changed from the highly cracked and porous surface characteristic for pure IrO₂ (x = 0) to more continuous and much less cracked PtO_y (x = 100).

All prepared mixed coupled $_xPtO_{y^-(100-x)}IrO_2$ electrodes showed the metallic character by supporting perfectly the reversible ferri/ferrocyanide reaction taken as the redox probe.

Voltammetric charges measured in the oxide region of potentials for all mixed coupled ${}_{x}PtO_{y^{-}(100-x)}IrO_{2}$ electrodes in 0.1 M KOH were found higher than for pure PtO_y. For mixed electrodes having the mol percentage of PtO_y equal or less than 50 % (x \leq 50), the voltammetric charge is found even higher than for pure IrO₂. The highest voltammetric charge was observed for the mixed of ${}_{40}PtO_{y^{-}60}IrO_{2}$ electrode, *i.e.* the electrode having PtO_y mol percentage of 40 % (x = 40). This could be ascribed to the increase of electrochemical surface area of the electrode due to the presence of both PtO_y and IrO₂.

In either basic (0.1 M KOH) or acid (0.1 M HClO₄) media, all mixed $_xPtO_{y^-(100-x)}IrO_2$ electrodes were found more electrocatalytically active for oxygen evolution reaction (OER) than pure PtO_y electrode. The mixed electrode having mol percentage of PtO_y equal or lower than 80 % (x \leq 80) showed similar, or for (x \leq 40) even better electrocatalytic activity for OER than pure IrO₂. Tafel slopes of mixed $_xPtO_{y^-(100-x)}IrO_2$ electrodes obtained from linear voltammograms corrected for ohmic drops, suggest faster OER on coupled mixed $_xPtO_{y^-(100-x)}IrO_2$ electrodes having (x \leq 80) than pure PtO_y. This suggests that just presence of IrO₂ determines high oxygen evolution rate at coupled mixed $_xPtO_{y^-(100-x)}IrO_2$ electrodes.

References

- [1] D. Gielen, F. Boshell, D. Saygin, *Nature Materials* **15** (2016) 117–120.
- [2] A. Bielecki, S. Ernst, W. Skrodzka, I. Wojnicki, *Research* **27** (2020) 11506–11530.
- [3] B. A. Olanipekun, N. O. Adelakun, *International Journal of Engineering Trends Technology*, 6**8(1)** (2020) 64–67.
- [4] S. Chu, A. Majumdar, *Nature* **488** (2012) 294–303.
- [5] M. K. Hubbert, *Science* **109** (1949) 103–109.
- [6] R. E. White, R. Grossman, *Science* **328** (2010) 5975173. <u>https://doi.org/10.1126/science.328.5975.173-a</u>.
- [7] T. Martin, Nature 538 (2016). <u>https://doi.org/10.1038/538171c</u>
- [8] J. Tollefson, Nature 521 (2015) 16-17. https://doi.org/10.1038/521016a
- [9] J. Mahmood, F. Li, S. Jung, M.S. Okyay, I. Ahmad, S. Kim, N. Park, H.Y. Jeong, J. Baek, Nature Nanotechnology 12(5) (2017) 441–446.
- [10] S. J. Gutić, A. S. Dobrota, E. Fako, N. V. Skorodumova, N. López, I. A. Pašti, *Catalysts* **10(3)** (2020) 290.
- [11] P. J. Rheinlander, J. Herranz, J. Durst, H. A. Gasteiger, *Journal of the Electrochemical Society*, 161(14) (2014) F1448–F1457.
- [12] Y. Xu, B. Zhang, Chemical Society Reviews 43 (2014) 2439–2450.
- [13] H.-J. Yin, J.-H. Zhou, Y.-W. Zhang, *Inorganic Chemistry Frontiers* **6** (2019) 2582-2618. <u>https://doi.org/-10.1039/C9QI00689C</u>
- [14] Z. Ma, Y. Zhang, S. Liu, W. Xu, L. Wu, Y.-C. Hsieh, P. Liu, Y. Zhu, K. Sasaki, J. N. Renner, K. E. Ayers, R. R. Adzic, J. X. Wang, Journal of Electroanalytical Chemistry 819(15) (2018) 296-305. <u>https://doi.org/10.1016/j.jelechem.2017.10.062</u>
- [15] K. V. Sankar, R. Attias, Y. Tsur, Electrochemistry Communications 110 (2020) 106641. <u>https://doi.org/-10.1016/j.elecom.2019.106641</u>
- [16] S-A Park, K.-S. Kim, Y.-T. Kim, ACS Energy Letters **3(5)** (2018) 1110–1115. <u>https://doi.org/10.1021/-acsenergylett.8b00368</u>
- [17] D. Majumdar, *Material Science Research India* **15(1)** (2018) 30–40.
- [18] S. Hadži-Jordanov, H. Angerstein-Kozlowska, M. Vuković, B. E. Conway, *Journal of the Electrochemical Society* **125** (1978) 1471–1480.
- [19] V. Birss, B. E. Conway, H. Angerstein-Kozlowska, Journal of the Electrochemical Society 131 (1984) 1502-1510.
- [20] D. Böhm, M. Beetz, M. Schuster, K. Peters, A. G. Hufnagel, M. Döblinger, B. Böller, T. Bein, D. Fattakhova-Rohlfing, Advanced Functional Materials 30 (2020) 1906670.
- [21] S. Li, C. Xi, Y.-Z. Jin, D. Wu, J.-Q. Wang, T. Liu, H.-B. Wang, C.-K. Dong, H. Liu, S. A. Kulinich, X.-W. Du, ACS Energy Letters 4(8) (2019) 1823–1829.

- [22] J. Cheng, J. Yang, S. Kitano, G. Juhasz, M. Higashi, M. Sadakiyo, K. Kato, S. Yoshioka, T. Sugiyama, M. Yamauchi, N. Nakashima, ACS Catalysis 9(8) (2019) 6974–6986.
- [23] J. Lee, B. Jeong, J. D. Ocon, *Current Applied Physics* **13** (2013) 309–321.
- [24] L. A. G. Pohan, L. Ouattara, La Revue Ivoirienne des Sciences et Technologie 19 (2012) 12–25.
- [25] J.-M. Hu, J.-Q. Zhang, C.-N. Cao, International Journal of Hydrogen Energy 29 (2004) 791–797.
- [26] O. Kambire, F. T. A. Appia, L. Ouattara, *La Revue Ivoirienne des Sciences et Technologie* 25 (2015) 21–33.
- [27] A. L. G. Pohan, L. Ouattara, K. H. Kondro, O. Kambiré, A. Trokourey, *European Journal of Scientific Research* **94(1)** (2013) 96–108.
- [28] O. Kambire, L. A. G. Pohan, F. T. A. Appia, C. Q.-M. Gnamba, K. H. Kondro, L. Ouattara, *Journal of Electrochemical Science and Engineering* **5(2)** (2015) 79–91.
- [29] A. L. G. Pohan, L. Ouattara, K. H. Kondro, O. Kambiré, A. Trokourey, *European Journal of Scientific Research* **94(1)** (2013) 96–108.
- [30] B. P. Hambly, J. B. Sheppard, B. D. Pendley, E. Lindner, *Electroanalysis* **30** (2018) 681–689.
- [31] H. Kahlert, U. Retter, H. Lohse, K. Siegler, F. Scholz, *Journal of Physical Chemistry B* **102(44)** (1998) 8757–8765.
- [32] A. J. Bard, L. R. Faulkner, *Electrochemical Methods: Fundamentals and Applications*, 2nd Edition, Wiley, 2001.
- [33] R. P. Janek, W. R. Fawcett, A. Ulman, *Langmuir* **14(11)** (1998) 3011–3018.
- [34] O. Kambire, L. A. G. Pohan, F. T. A. Appia, C. Q.-M. Gnamba, K. H. Kondro, L. Ouattara, *Journal of Electrochemical Science and Engineering* **5(2)** (2015) 79–91.
- [35] T. A. F. Lassali, S. C. De Castro, J. F. C. Boodts, *Electrochimica Acta* **43 (16-17)** (1998) 2515–2525.
- [36] K. H. Kondro, L. Ouattara, A. Trokourey, Y. Bokra, *Bulletin of the Chemical Society of Ethiopia* **22(1)** (2008) 125–134.
- [37] A. Kapałka, G. Fóti, C. Comninellis, *Electrochemistry Communications* **10(4)** (2008) 607–610.
- [38] N. Katsuki, E. Takahashi, M. Toyoda, T. Kurosu, M. Iida, S. Wakita, Y. Nishiki, T. Shimamune, *Journal of the Electrochemical Society* **145(7)** (1998) 2358–2362
- [39] G. H. Kelsall, Journal of Applied Electrochemistry **14(2)** (1984) 177–186
- [40] O. Kambire, L. A. G. Pohan, F. T. A. Appia, L. Ouattara, *International Journal of Pure and Applied Science* & *Technology* **27(1)** (2015) 27-43.
- [41] B. M. Jović, U.Č. Lačnjevac, V.D. Jović, N.V. Krstajić, *Journal of Electroanalytical Chemistry* **754** (2015) 100–108.
- [42] B. M. Jović, U. Č. Lačnjevac, V. D. Jović, L. Gajić-Krstajić, J. Kovač, D. Poleti, N. V. Krstajić, *International Journal of Hydrogen Energy* **41** (2016) 20502–20514.
- [43] S. Trasatti, in: Encyclopedia of Electrochemical Power Sources, 1st Edition, J. Garche, C. Dyer, P. Moseley,
 Z. Ogumi, D. Rand, B. Scrosati (Eds.), Elsevier Science, Amsterdam, Vol. 2, 2009, pp. 49–55.

©2020 by the authors; licensee IAPC, Zagreb, Croatia. This article is an open-access article distributed under the terms and conditions of the Creative Commons Attribution license (https://creativecommons.org/licenses/by/4.0/)